

Additive Manufacturing of Metals and Alloys

8. Post-treatments for microstructure and properties control

February 2023

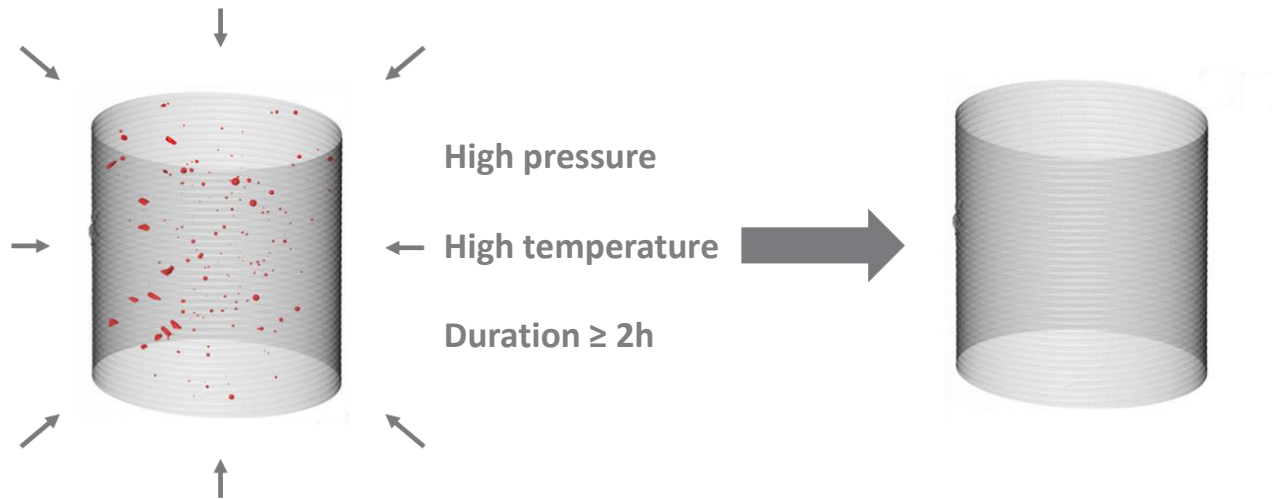
Dr. Charlotte de Formanoir
Prof. Roland Logé
MER Dr. Christian Leinenbach



Post-treatments for microstructure and properties control

- HIP
 - process conditions
 - limitations
 - microstructural effects
- Stress-relief
 - process conditions
 - microstructural effects
- Thermal post-treatments for microstructure control
 - typical microstructural effects
 - influencing factors (illustrated on Ti-6Al-4V)
- Surface post-treatments
- Effect of post-processing on mechanical properties
 - tensile
 - fatigue

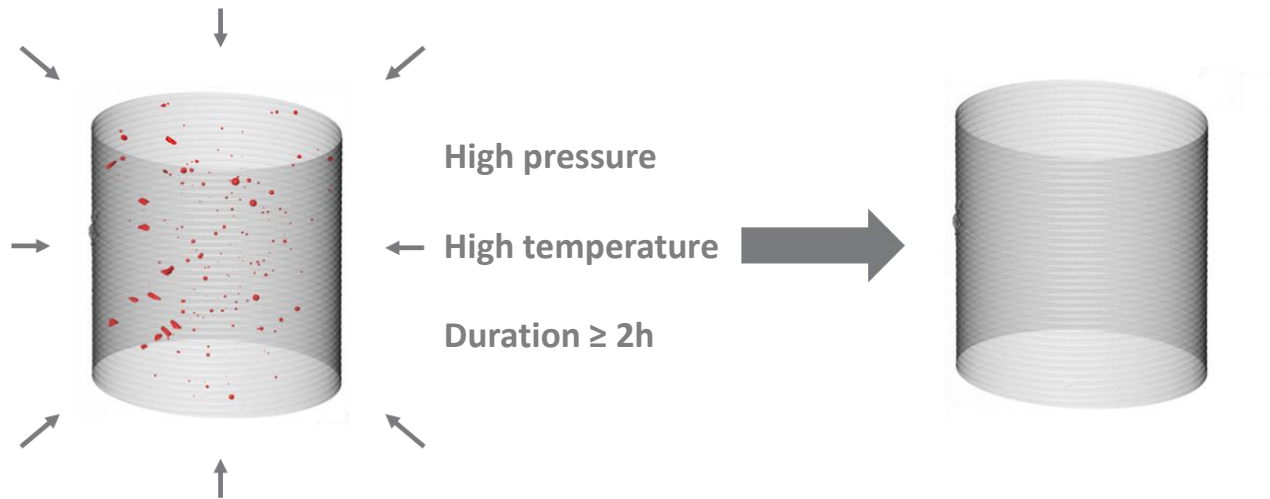
Hot Isostatic Pressing (HIP)



HIP is a thermomechanical post-treatment allowing to **remove pores** entrapped in the as-built AM material.

Post-HIP relative densities **typically exceed 99.9%.**

HIP: process conditions



HIP parameters (duration, temperature, pressure) are **material-dependent**.

Material		Temperature	Pressure	Duration
Ni-based alloys	IN718	1200°C	120 MPa	4h
		1180°C	150 MPa	3h
		1150°C	100 MPa	4h
	IN625	1120–1240 °C	100–165 MPa	3-4h
	Hastelloy X	1155°C	100 MPa	3h
Ti alloys	Ti-6Al-4V	900°C – 920°C	100 MPa	2h
Al alloys	AlSi10Mg	500°C	100 MPa	2h
Mg alloys	AZ61	450°C	103 MPa	3h
Stainless steel	316L	1125°C	137 MPa	4h
		1150°C	100 MPa	4h
	17-4 PH	1120°C	105 MPa	2h
CoCr alloys	CoCrMo	1200°C	100 MPa	4h

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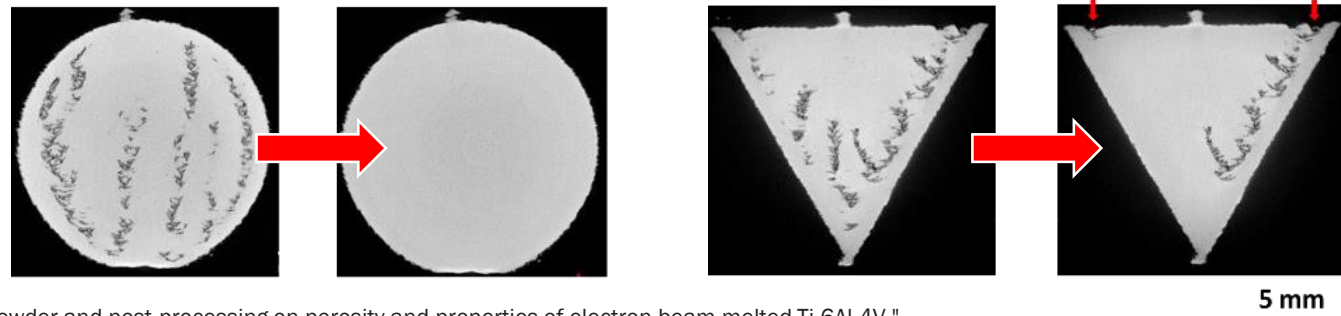
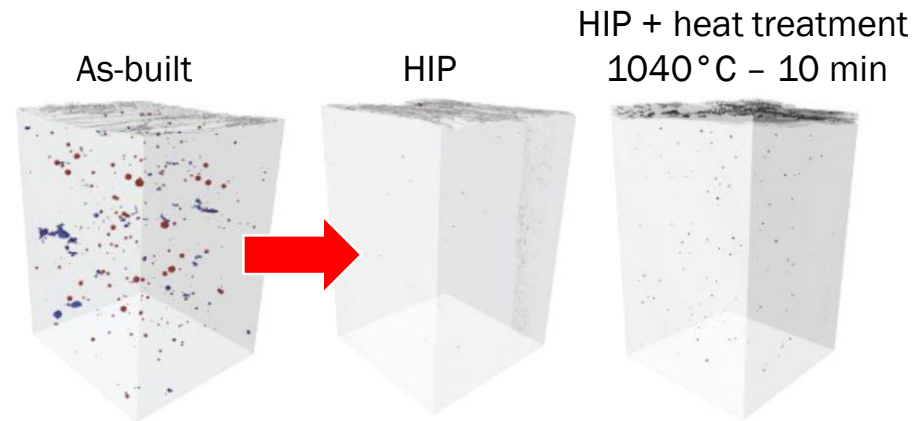
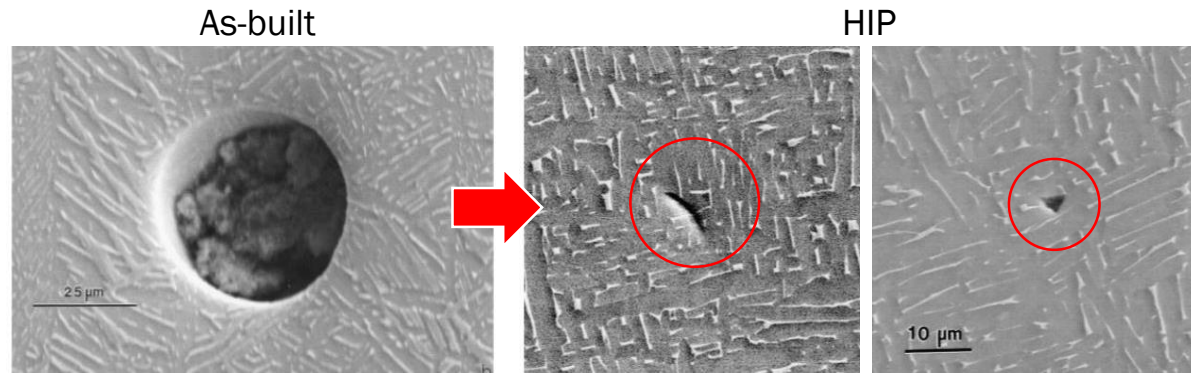
HIP: limitations

Remnants = small argon-filled collapsed voids.

HIP induces a radius shrinkage of 1 to 2 orders of magnitude, resulting in a modest to significant increase in the **internal pressure** of the **Ar-containing pores**, which impairs their full closure.

Regrowth

Ineffective for open porosity



Cunningham, Ross, et al. "Analyzing the effects of powder and post-processing on porosity and properties of electron beam melted Ti-6Al-4V." *Materials Research Letters* 5.7 (2017): 516-525.

Tammas-Williams, Samuel, et al. "The effectiveness of hot isostatic pressing for closing porosity in titanium parts manufactured by selective electron beam melting." *Metallurgical and Materials Transactions A* 47.5 (2016): 1939-1946.

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HIP: microstructural effect

Al-based alloys (AlSi10Mg)

Melt pool boundaries are no longer visible after HIP.

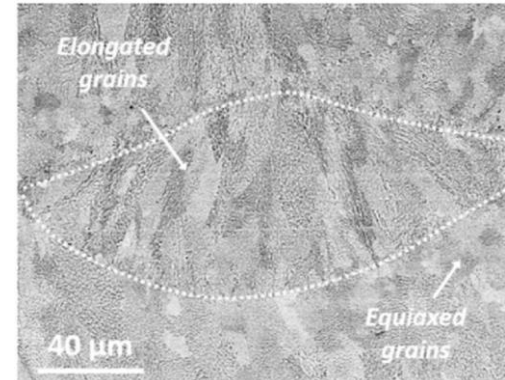
Prior to HIP:

fine, sub-micron cellular eutectic microstructure, elongated along the build direction

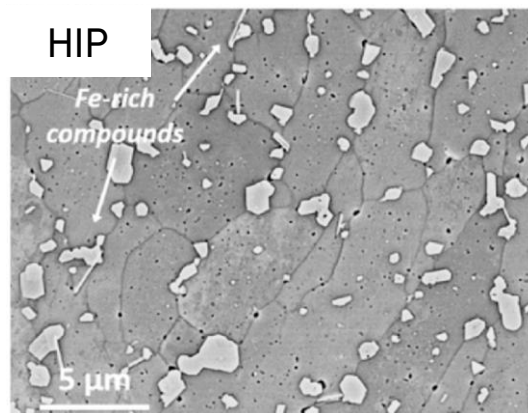
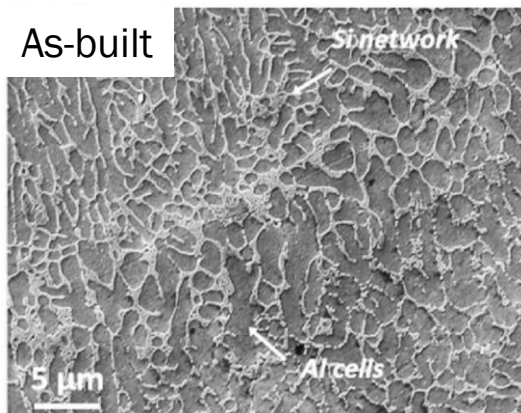
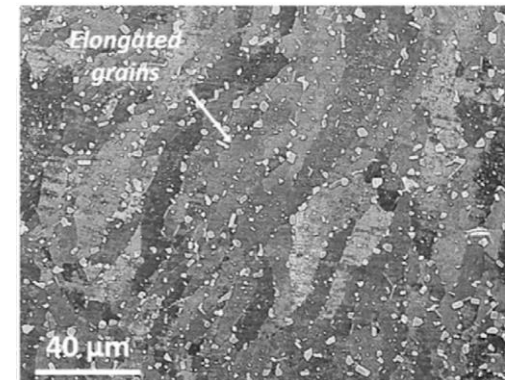
Post HIP:

diffusion and agglomeration of Si into large Si particles

As-built



HIP

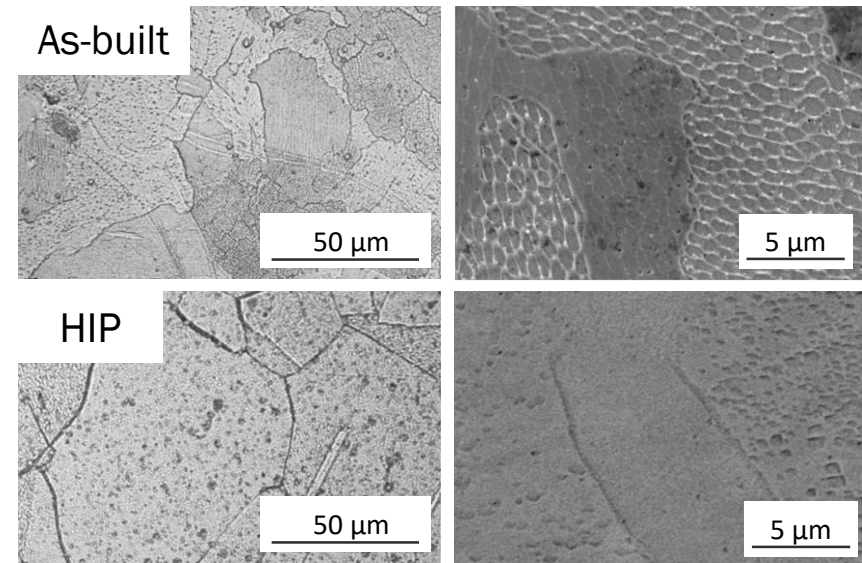


HIP: microstructural effect

316L

Prior to HIP: cellular microstructure

Post HIP: large equiaxed grains.

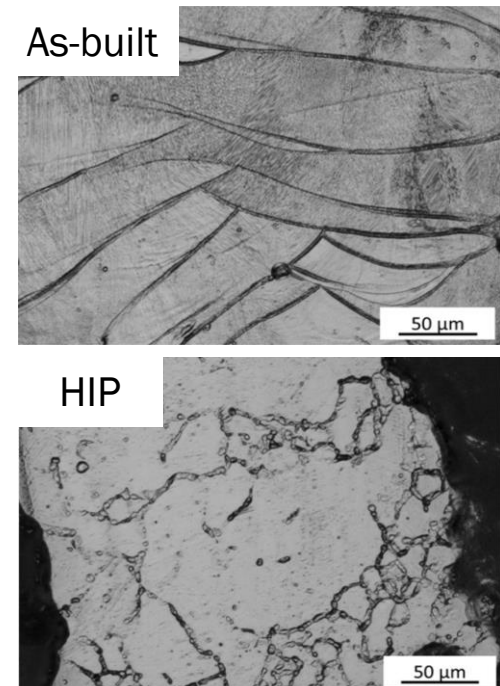


CoCrMo

Prior to HIP: cellular microstructure

Post HIP: large equiaxed grains.

Melt pool boundaries are no longer visible after HIP.



Montero-Sistiaga, Maria L., et al. "Microstructure evolution of 316L produced by HP-SLM (high power selective laser melting)." *Additive Manufacturing* 23 (2018): 402-410.

A. Cutolo, et al. "Influence of layer thickness and post-process treatments on the fatigue properties of CoCr scaffolds produced by laser powder bed fusion", *Addit. Manuf.* 23 (2018) 498-504.

Kreitchberg, Alena, et al. "Effect of heat treatment and hot isostatic pressing on the microstructure and mechanical properties of Inconel 625 alloy processed by laser powder bed fusion." *Materials Science and Engineering: A* 689 (2017): 1-10.

HIP: microstructural effect (global)

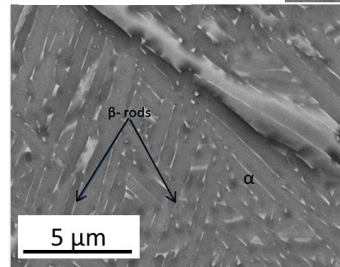
Ti-6Al-4V

EBM/DED: [microstructural coarsening](#)

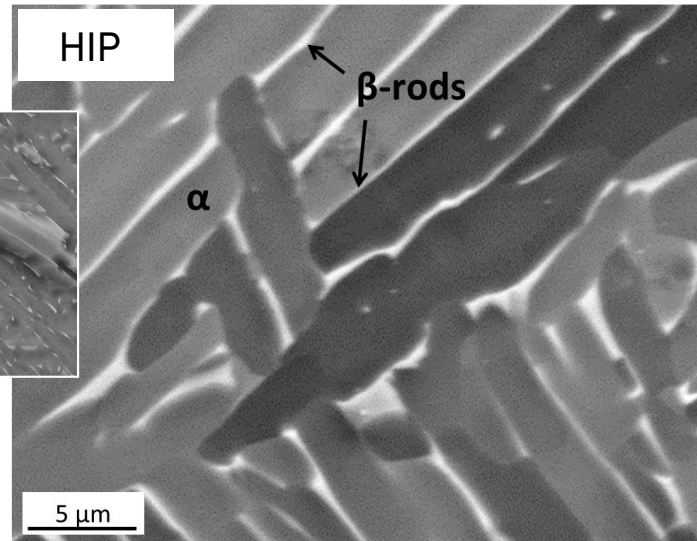
α lath thickness increases from 1-2 μm in as-built condition to 3-5 μm after HIP

EBM

As-built

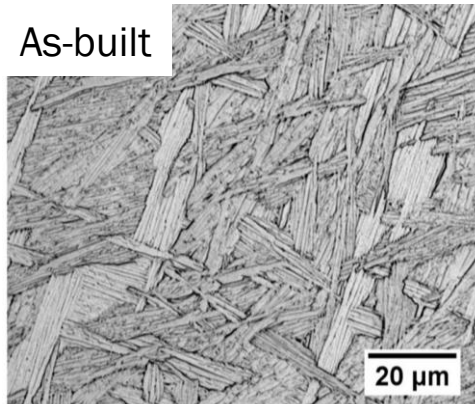


HIP

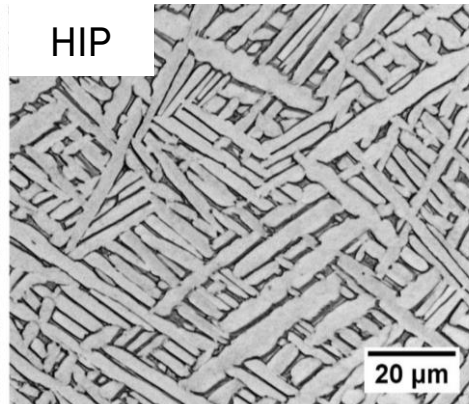


DED

As-built



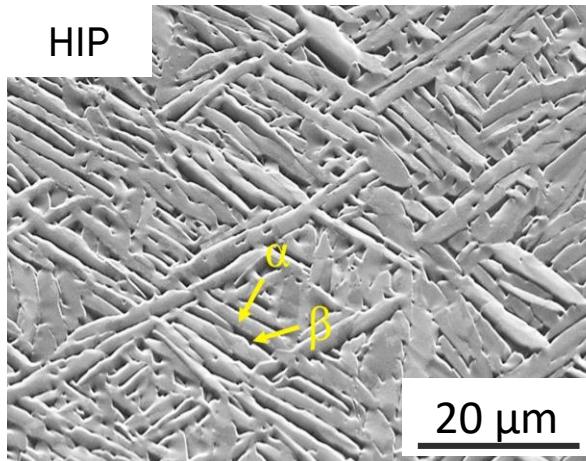
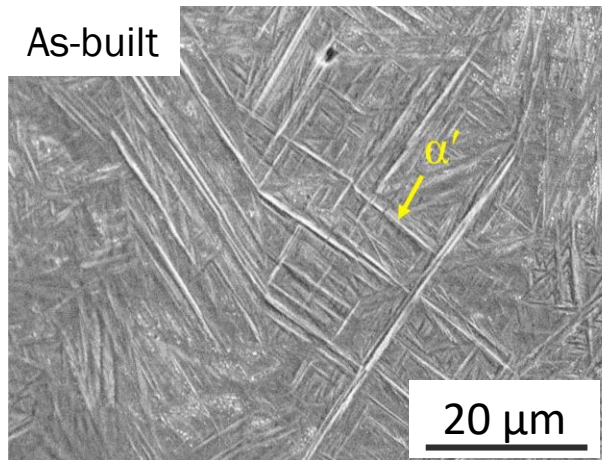
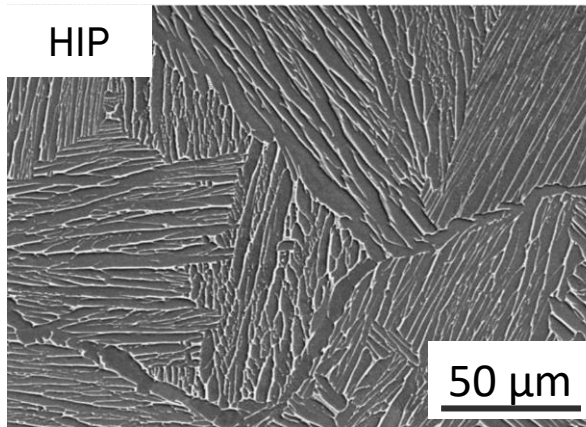
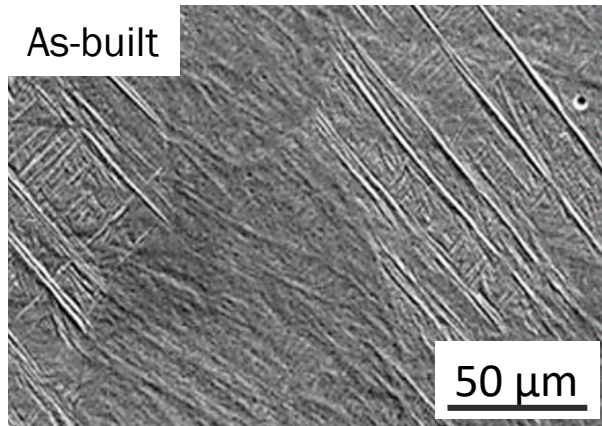
HIP



HIP: microstructural effect (global)

Ti-6Al-4V

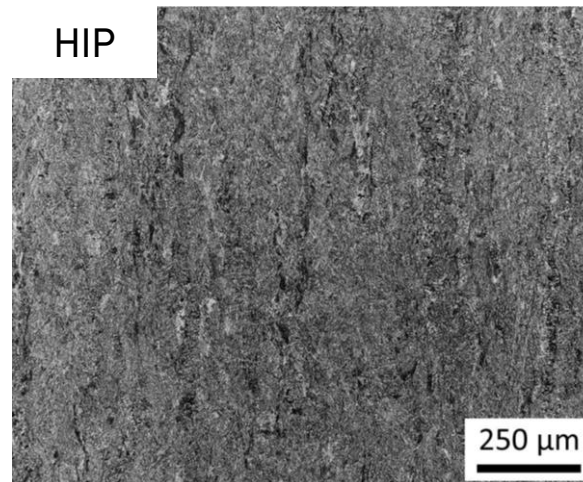
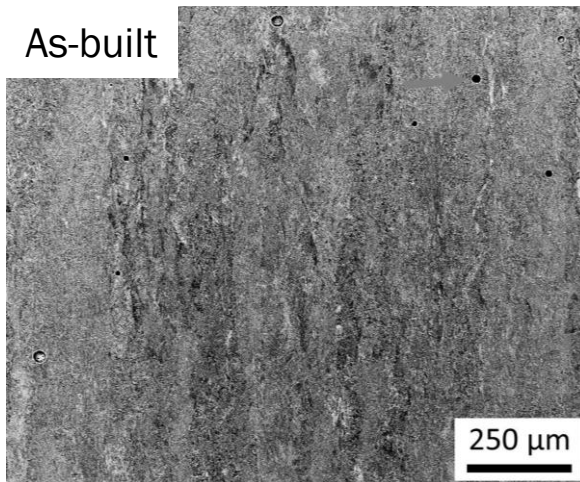
LPBF: martensite decomposition $\alpha' \rightarrow \alpha + \beta$ and microstructural coarsening
 α lath thickness of about 4 μm after HIP



HIP: microstructural effect (global)

Ti-6Al-4V

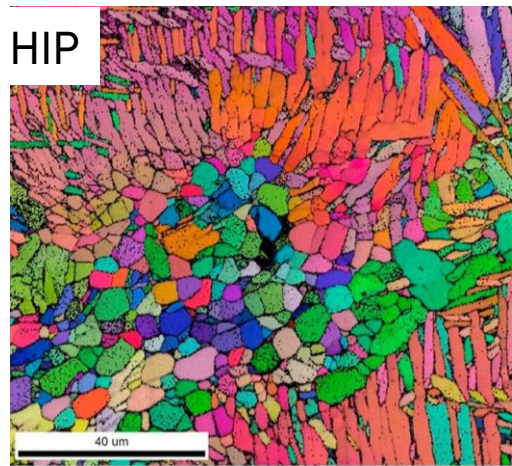
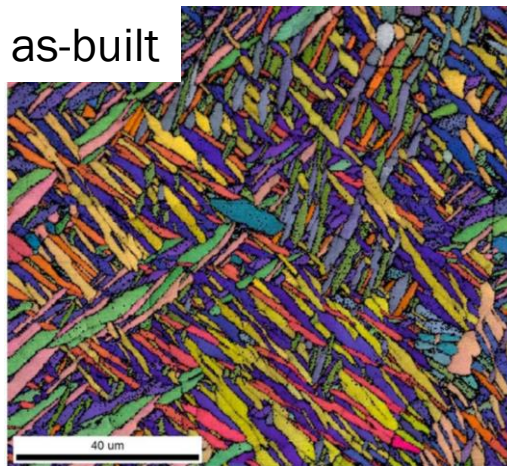
HIP has **no effect on parent β grains** columnar morphology (because HIP is performed **below the transus temperature**).



HIP: microstructural effects (in the vicinity of pores)

Ti-6Al-4V

A fine equiaxed microstructure is observed at the locations of former defects.



High-temperature deformation takes place **near voids** during HIP:
local hot working, with substantial deformation in the $\alpha+\beta$ temperature range.
Other locations of the specimen undergo less deformation during compaction.
 \Rightarrow more refined post-HIP microstructure near defects

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Stress-relief: process conditions

Stress relief SR: thermal post-treatment allowing to **reduce or eliminate residual stresses**.

Objective: **decrease distortions** before cutting the parts from the building platform, without significantly affecting the grain structure.

1. **uniform heating** to a suitable temperature below the transformation range,
2. holding at this temperature for a given period of time,
3. **uniform cooling**

Material	Temperature
Co-Cr-Mo	450 °C/45min + 750 °C/1h/furnace cooling FC
Ni-based alloys (IN625, IN718)	650-870 °C/1-8h
Steels (316L, AISI 4340)	470-650 °C/40min-5h
Ti-6Al-4V	600-750 °C/2-3h/furnace cooling or air cooling
Al alloys (4xxx serie: Al-Si)	150-400 °C/2-6h/furnace cooling or air cooling

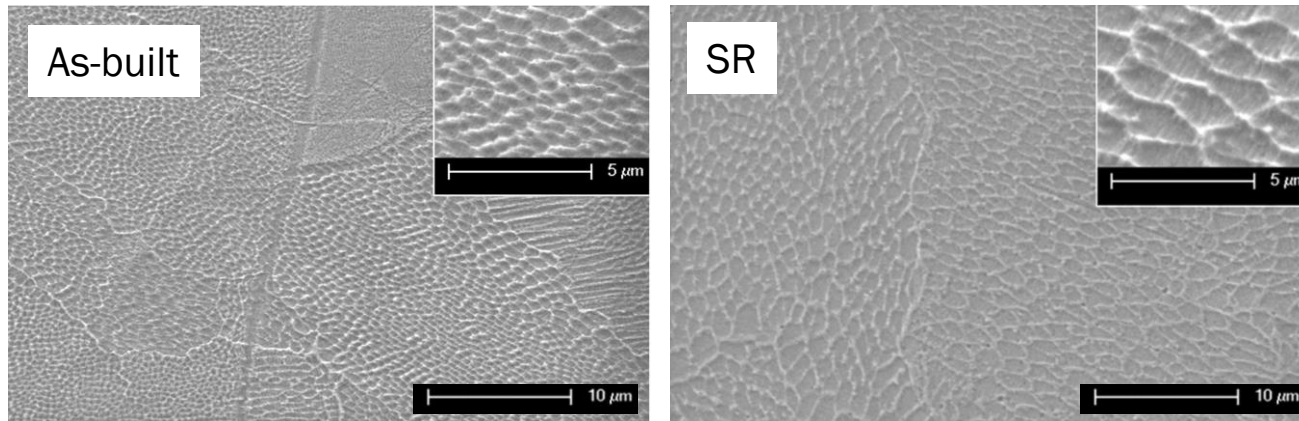
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Stress-relief: microstructural effect

In some cases (e.g. steels), the **SR temperature** is **too low** for major microstructural changes to occur.

ex: **316L** – SR at 470 °C – slight coarsening of the cellular solidification morphology

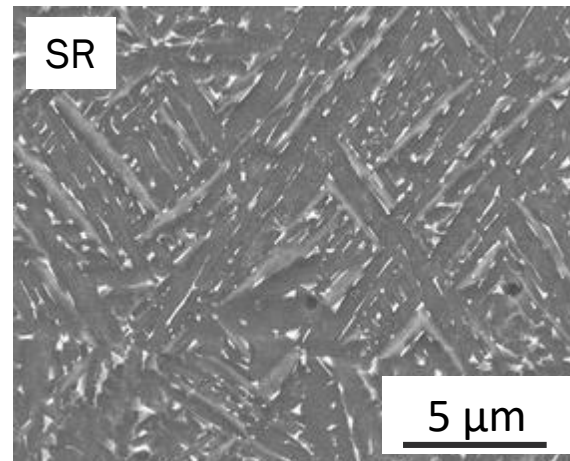
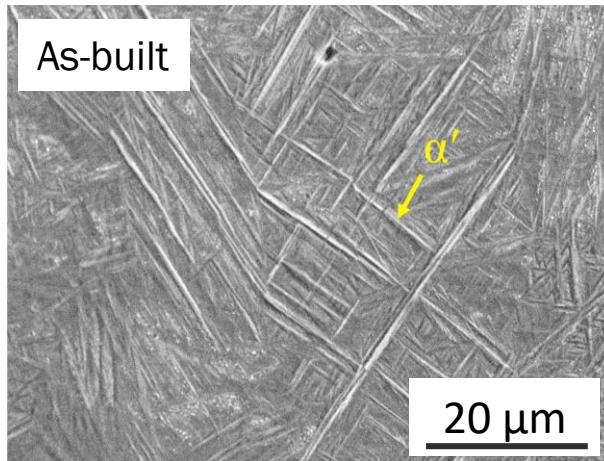


Condition	$\sigma_{y,0.2\%}$ [MPa]	UTS [MPa]	Elongation before fracture [%]
As-built	453 ± 6	573 ± 6	46 ± 1
Stress relief heat treated	449 ± 5	570 ± 5	48 ± 1

Stress-relief: microstructural effect

In other cases (e.g. Al alloys, Ti64) the SR treatment can have a more significant effect

ex: **Ti64** – SR at 730 °C/2h – decomposition of the martensite $\alpha' \rightarrow \alpha + \beta$



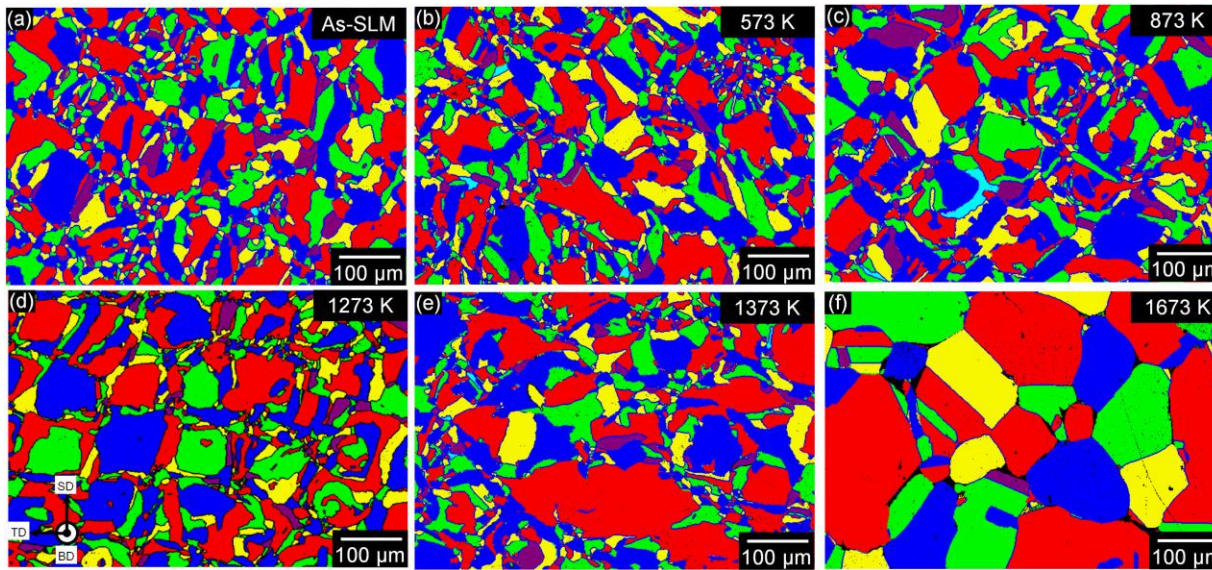
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Heat treatments: typical microstructural effects

Grain growth

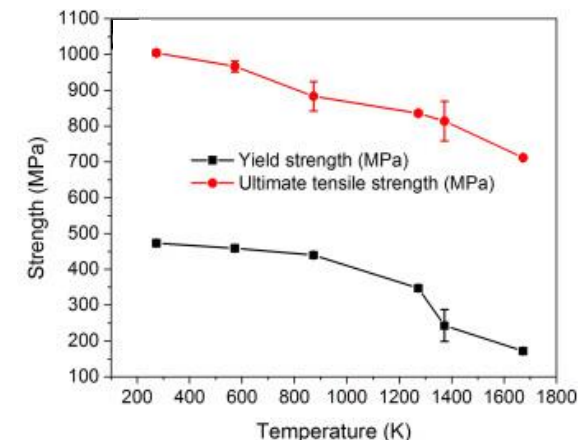
ex: 316L



The grains become more **equiaxed** and **larger** with increasing annealing temperature.

Average grain size

as-built: $45 \pm 1 \mu\text{m}$
300 °C: $50 \pm 3 \mu\text{m}$
600 °C: $55 \pm 3 \mu\text{m}$
1000 °C: $65 \pm 8 \mu\text{m}$
1100 °C: $88 \pm 5 \mu\text{m}$
1400 °C: $102 \pm 3 \mu\text{m}$



Salman, O. O., et al. "Effect of heat treatment on microstructure and mechanical properties of 316L steel synthesized by selective laser melting." *Materials Science and Engineering: A* 748 (2019): 205-212.

Heat treatments: typical microstructural effects

Recrystallization

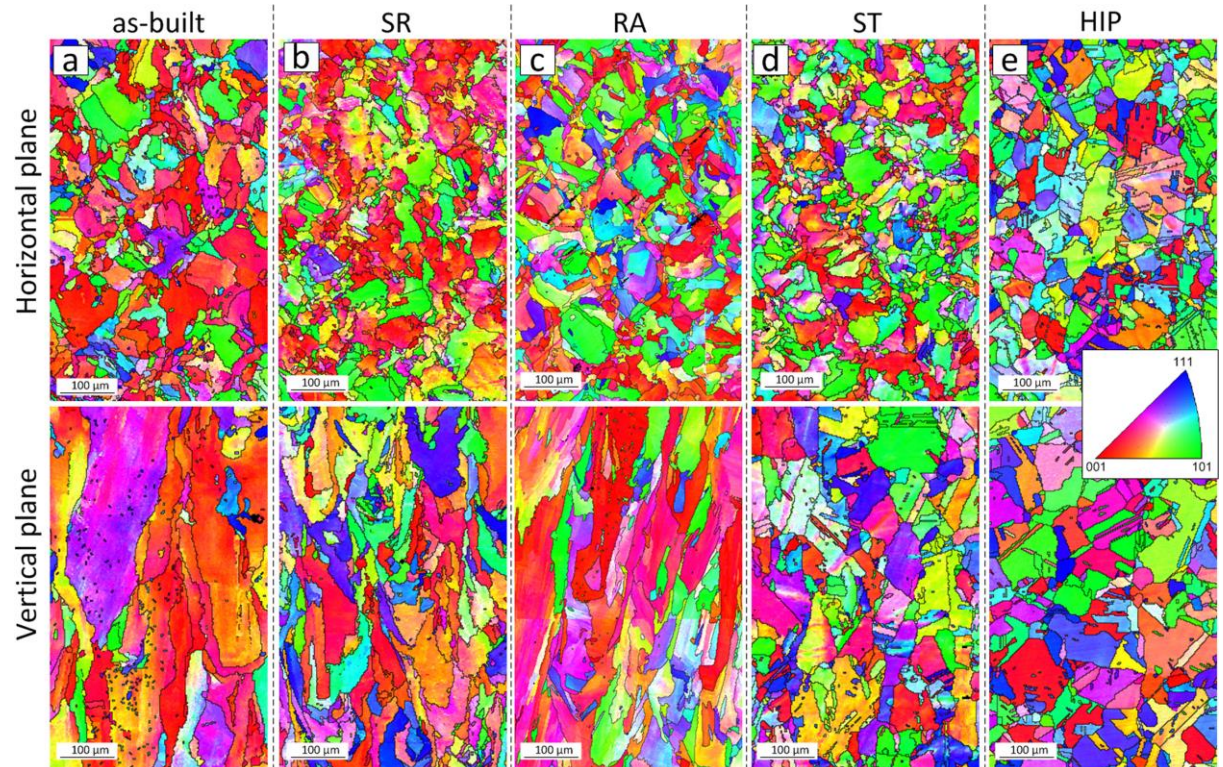
ex: IN625

SR: 650-870°C

RA: 930-1040°C

ST: 1040-1200°C

HIP: 1120-1240°C



Heat treatments: typical microstructural effects

Recrystallization

ex: IN625

As-built:

bimodal misorientation angle distribution

⇒ mixed grain/subgrain structure

SR:

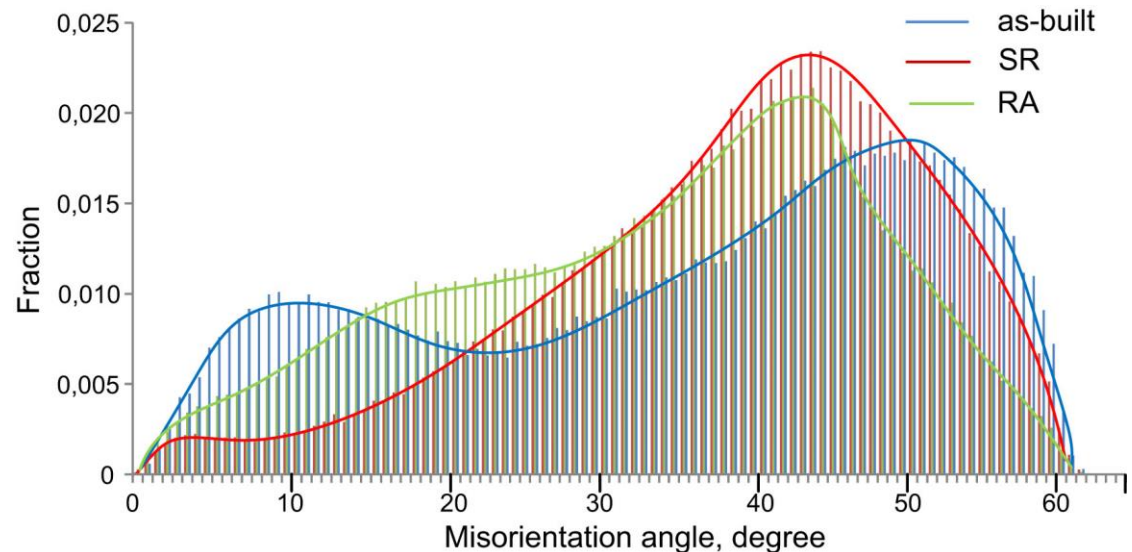
The fraction of low-angle grain boundaries decreases, while the fraction of high-angle grain boundaries increases

⇒ coalescence of subgrains

RA:

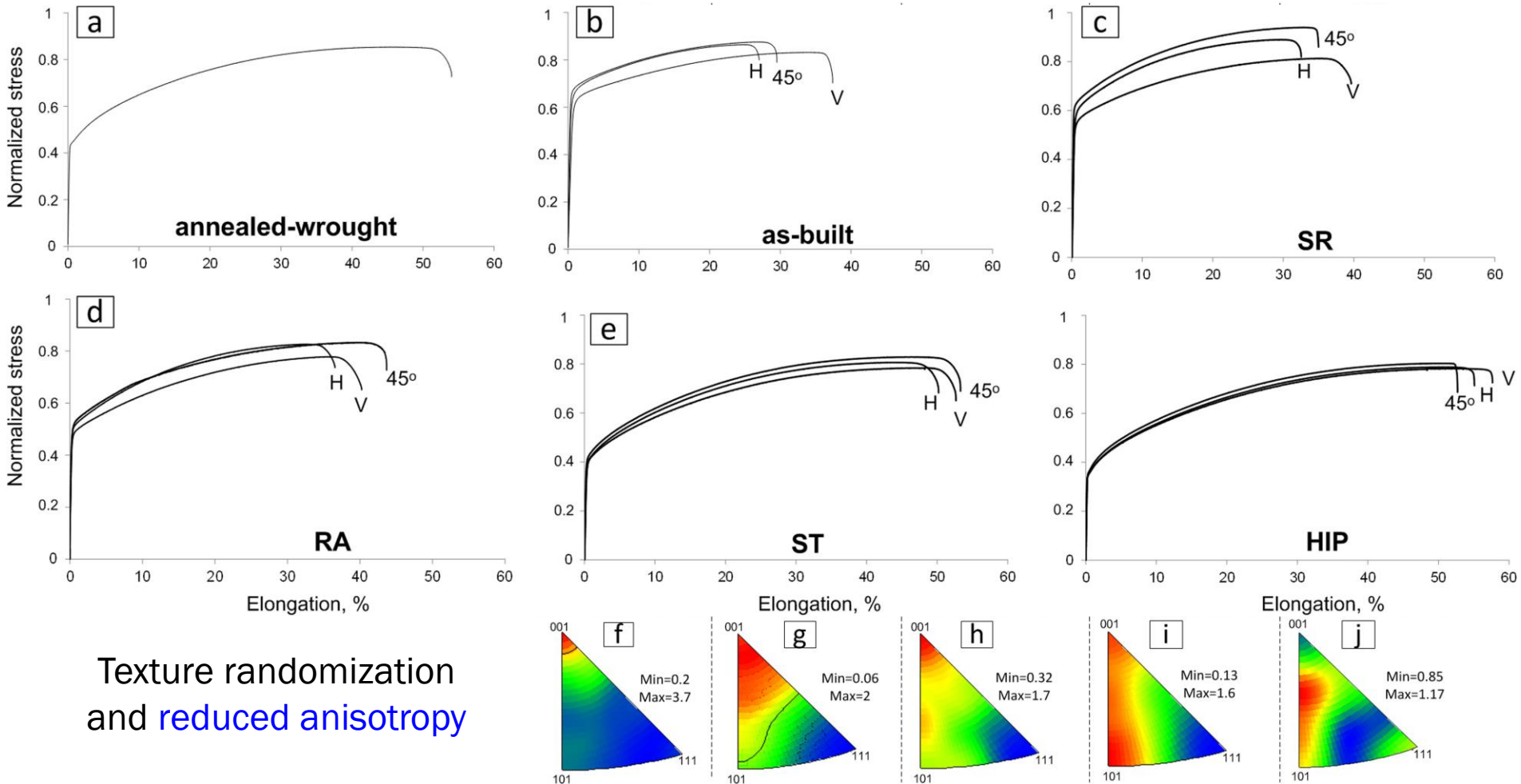
The amount of grain boundaries with 15–30° misorientation increases

⇒ partial recrystallization



Heat treatments: typical microstructural effects

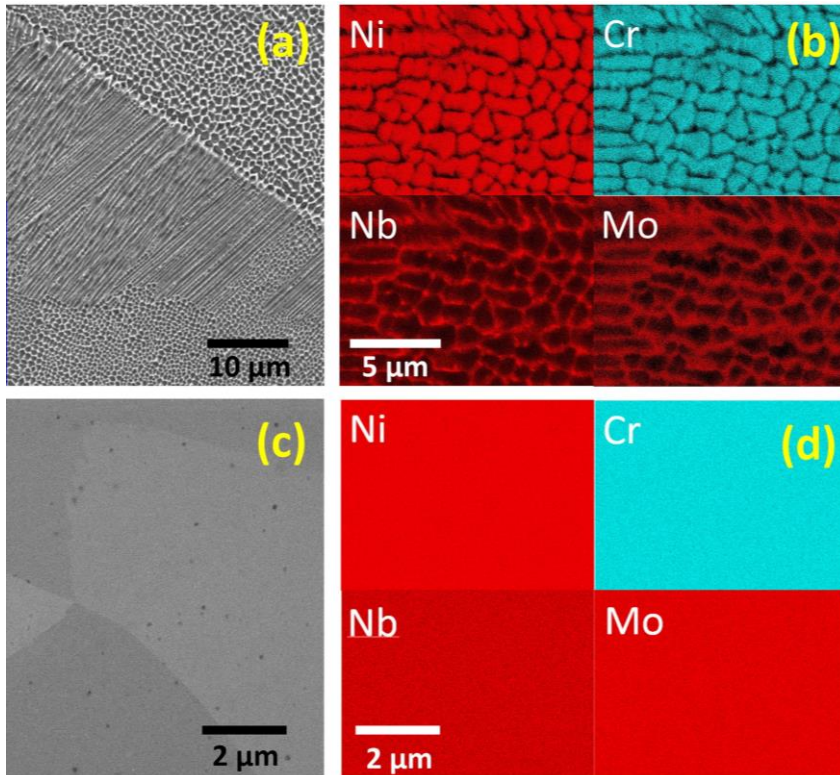
Recrystallization



Heat treatments: typical microstructural effects

Microstructure homogenization

ex: IN625



As-built:

fine dendritic microstructures and **elemental segregation** due to rapid solidification

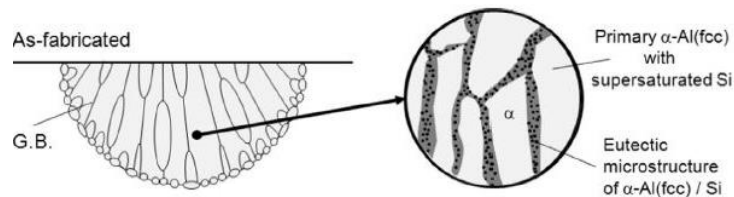
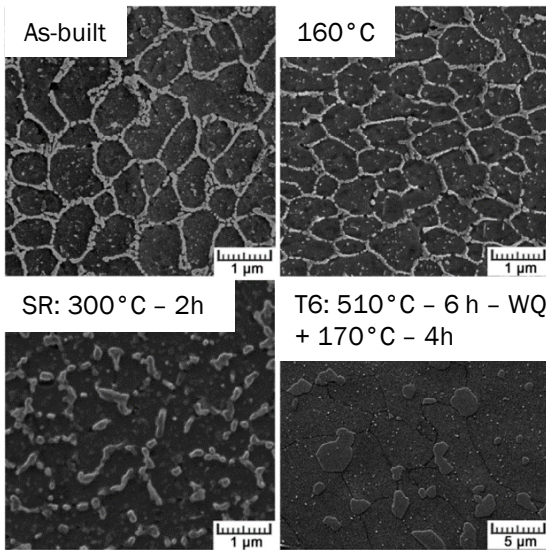
ST - 1150 °C - 1 h:

uniform chemical composition

Heat treatments: typical microstructural effects

Precipitation

ex: Al-Si alloys (4xxx series)



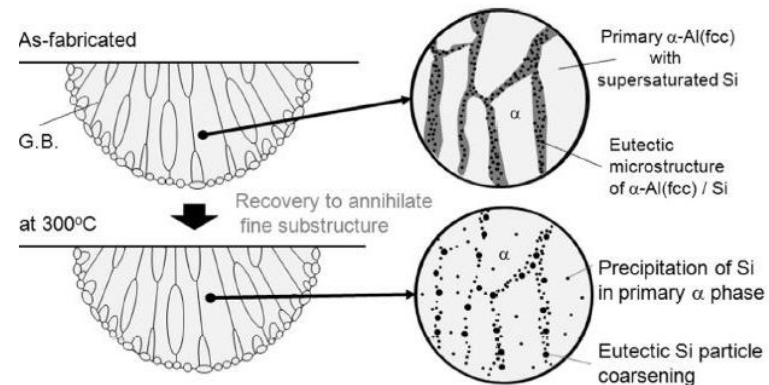
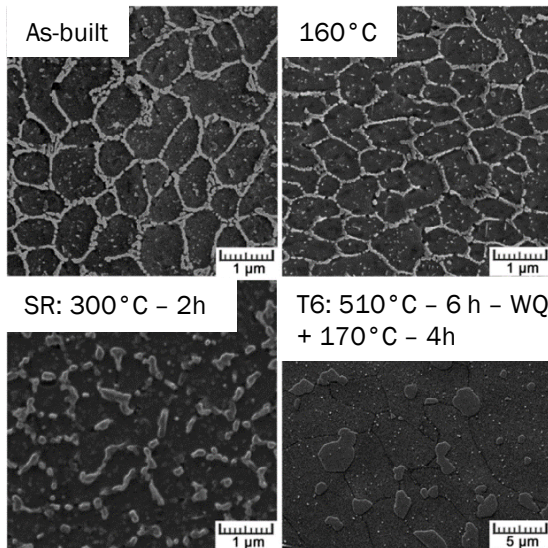
As-built cellular microstructure: primary α -Al columnar grains with a **Si-rich network at the cell boundaries**.

At annealing temperatures **as low as 160°C**: Precipitation of **nano-sized pure Si in the matrix** but unchanged intercellular Si-rich network.

Heat treatments: typical microstructural effects

Precipitation

ex: Al-Si alloys (4xxx series)



At 300°C: breaking up of the Si network acting as nucleation sites for the formation of Si particles

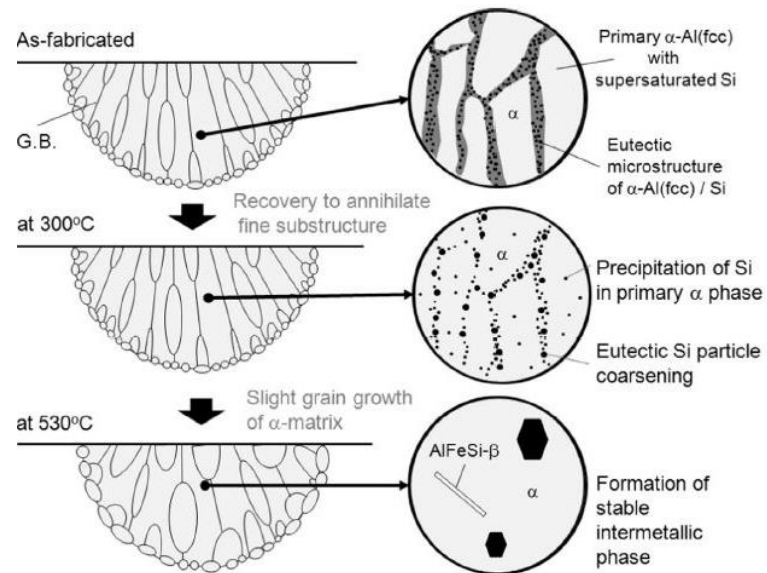
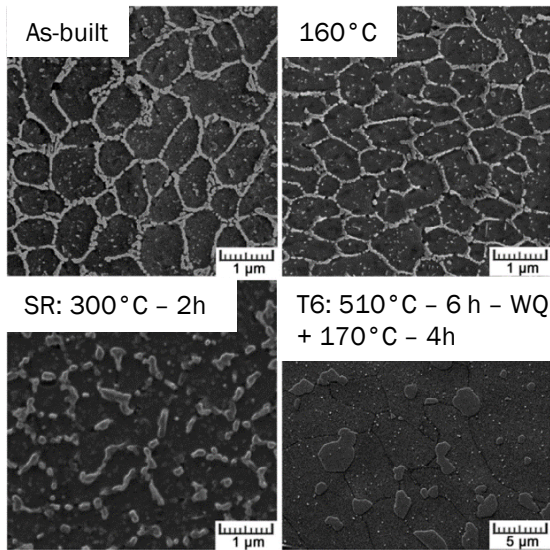
Diffusion of excess Si from the α -Al matrix

⇒ coarsening of the Si particles at these nucleation sites

Heat treatments: typical microstructural effects

Precipitation

ex: Al-Si alloys (4xxx series)



Above 500°C: further grain growth, Si phase coarsening and formation of a stable iron intermetallic phase

Heat treatments: typical microstructural effects

Precipitation

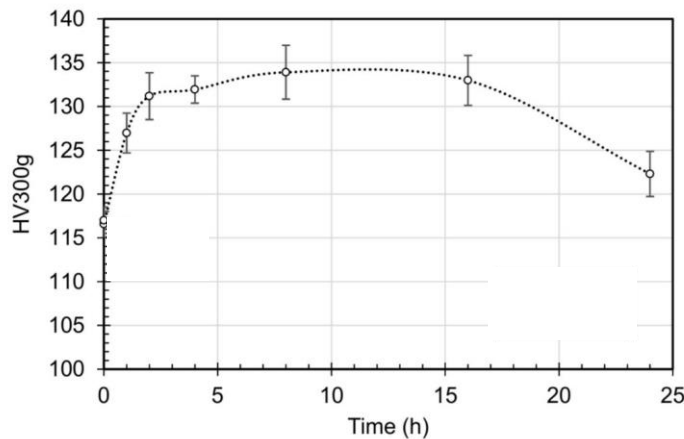
ex: Al-Si alloys (4xxx series) – **direct aging** T5

Skip the solution-annealing stage (first step of the T6 treatment) which induces grain growth and coarsening of the Si particles

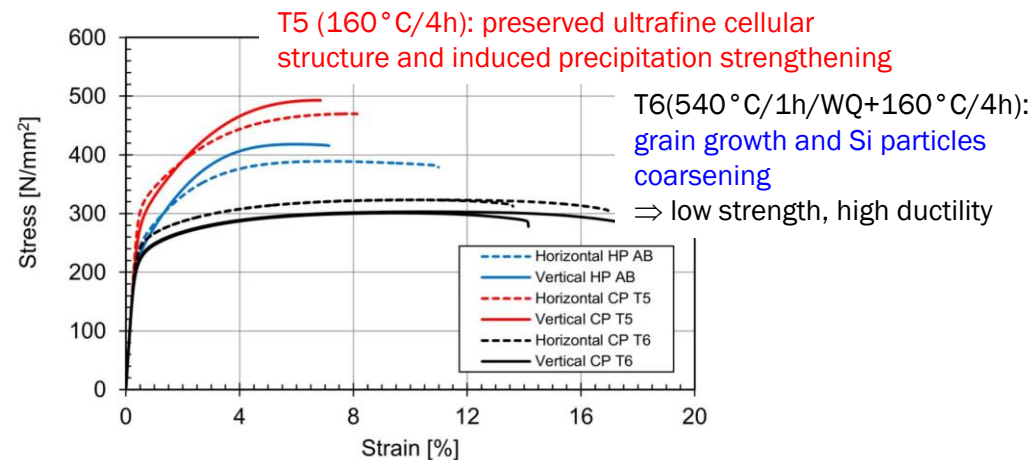
⇒ Promote **precipitation aging** while starting right from the as-built material

⇒ **Exploit**

- **supersaturation** of the solid solution to enhance precipitation strengthening
- **fine cellular structure** to maximize strengthening



Vickers' microhardness evolution as a function of holding time at the **aging** temperature of 160°C

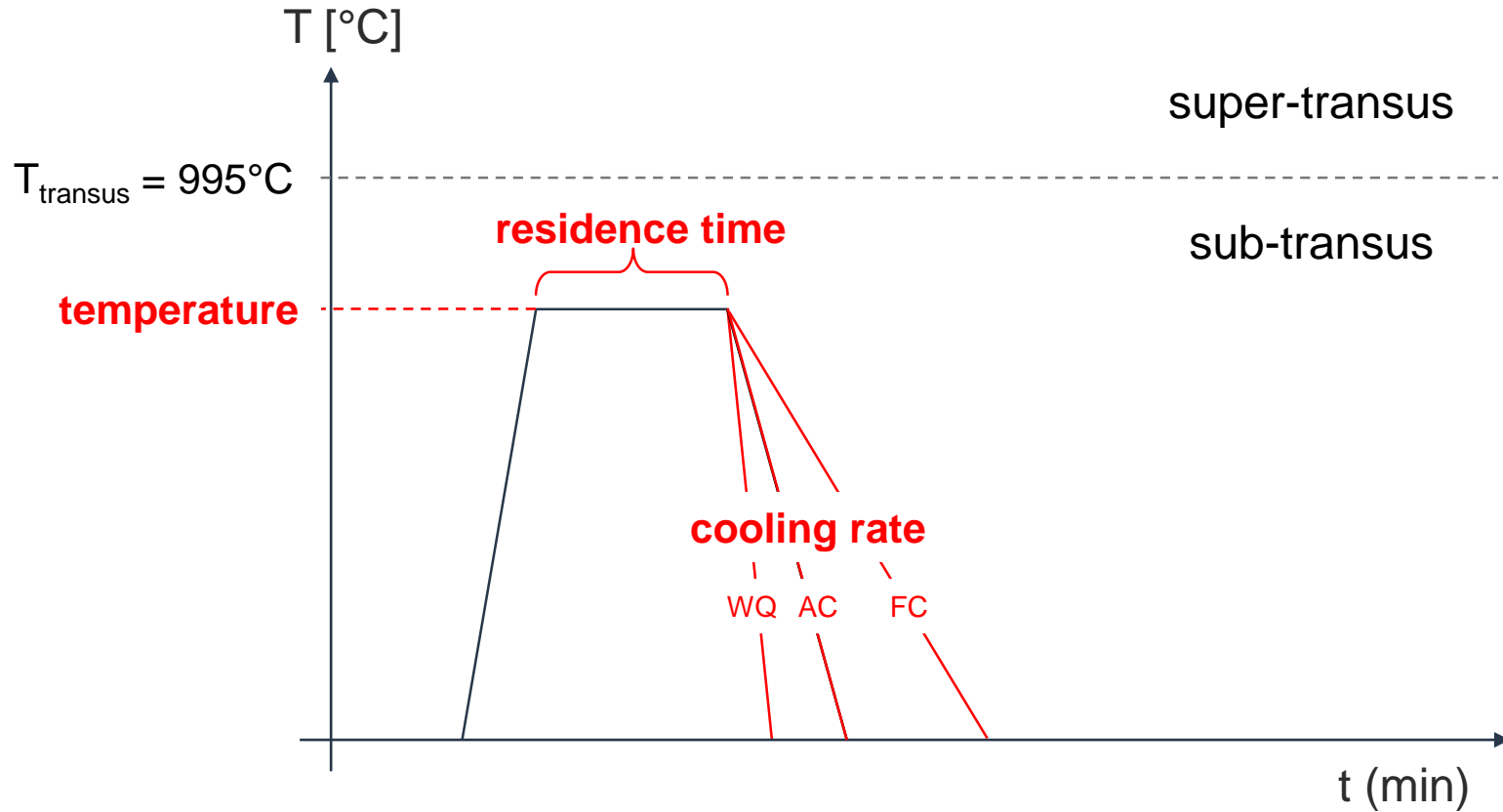


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Thermal post-treatments of Ti-6Al-4V

Effect of heat treatments: **influencing factors**

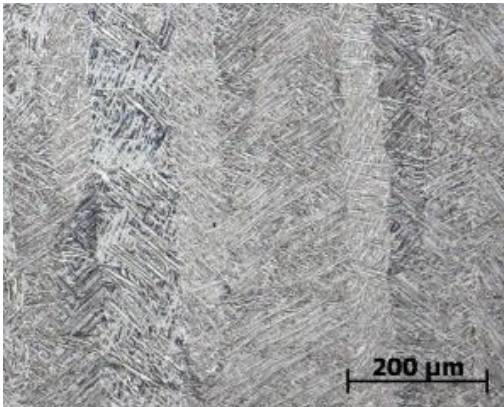


WQ : water quenching
AC : air cooling
FC : furnace cooling

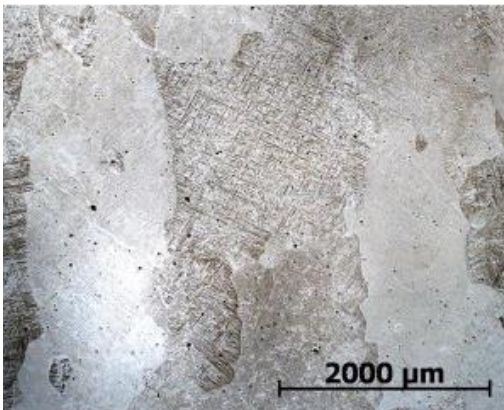
Thermal post-treatments of Ti-6Al-4V

Effect of heat treatments: influencing factors

Temperature



Sub-transus temperature (940° C):
the columnar microstructure is maintained,
the growth of the β grain is hindered by the
presence of α phase along the β grain
boundaries



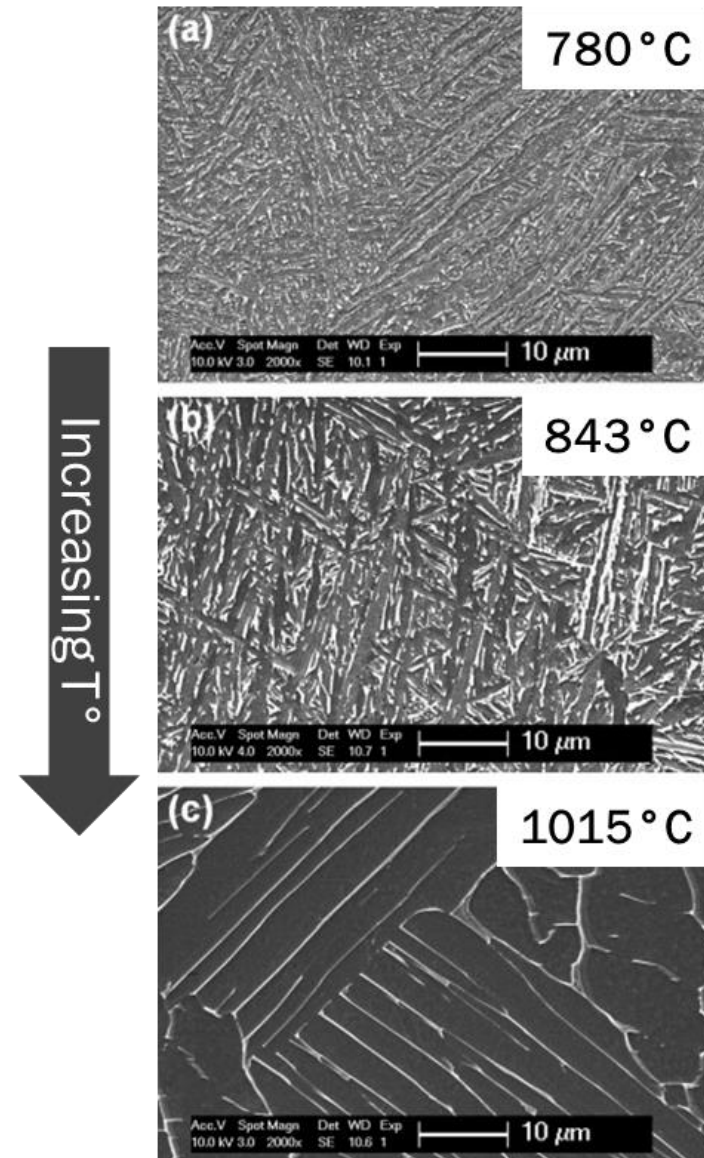
Super-transus temperature (1015° C):
uncontrollable β grain growth (several mm)

Thermal post-treatments of Ti-6Al-4V

Effect of heat treatments: influencing factors

Temperature

At high temperatures, **martensite decomposition** $\alpha' \rightarrow \alpha + \beta$ and **microstructural coarsening** result in coarse α lamellae delineated by β phase.

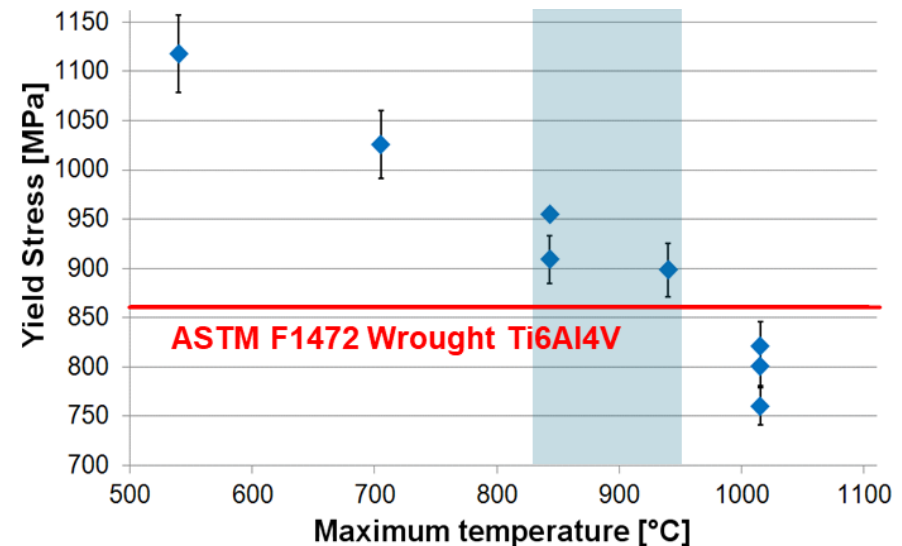
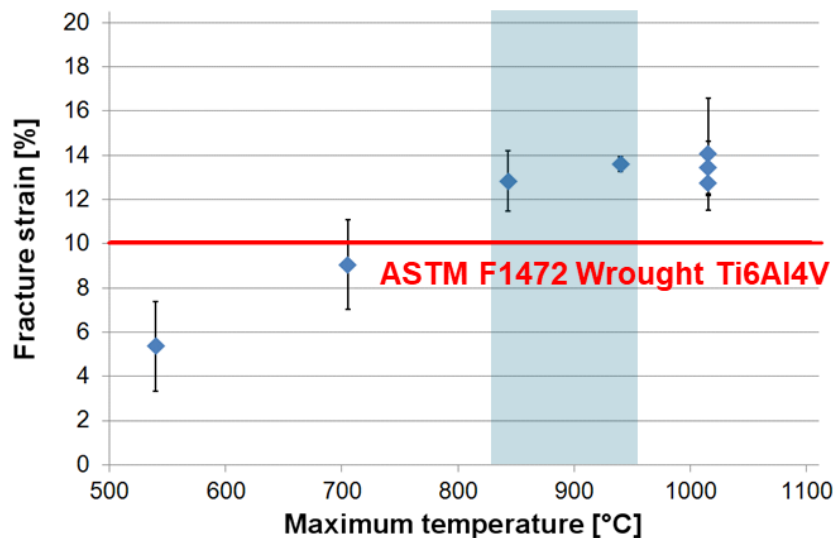


Thermal post-treatments of Ti-6Al-4V

Effect of heat treatments: influencing factors

Temperature

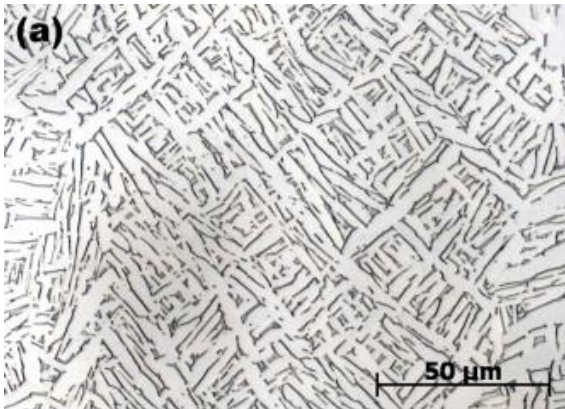
Martensite decomposition substantially increases the **ductility** of the material but reduces the **strength**. For LPBF Ti6Al4V, the **optimal** heat treatment temperature ranges **between 850 °C and 950 °C**.



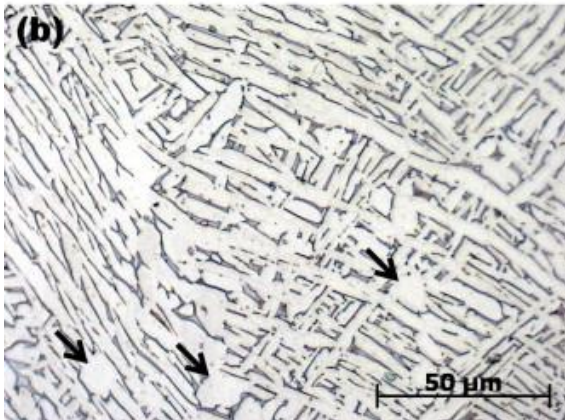
Thermal post-treatments of Ti-6Al-4V

Effect of heat treatments: influencing factors

Residence time



2h at 940 °C



20h at 940 °C

At **sub-transus** temperatures, the **residence time** has a very **limited influence** on the microstructure.

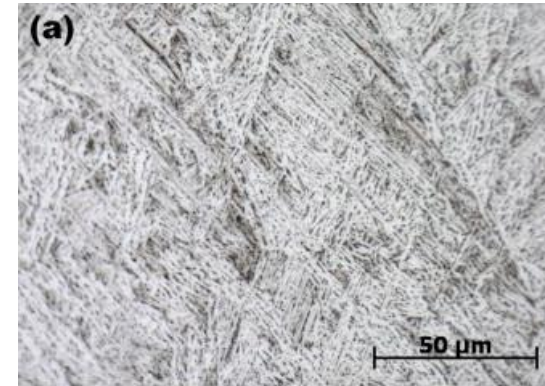
Thermal post-treatments of Ti-6Al-4V

2h at 850°C

Effect of heat treatments: influencing factors

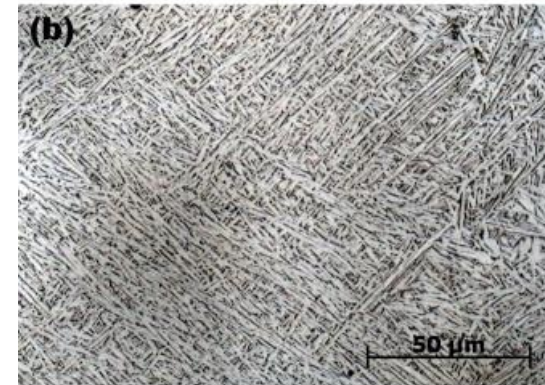
Cooling rate

Furnace cooling



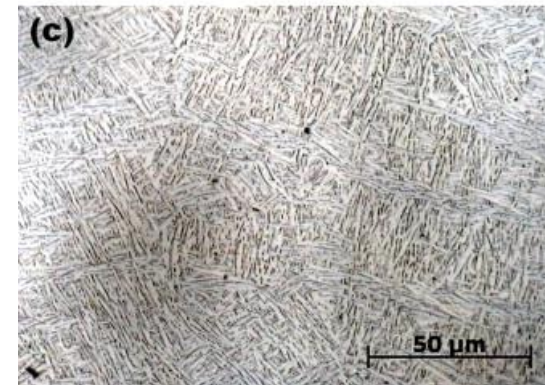
The influence of the cooling rate on the resulting microstructure increases as the annealing temperature increases.

Air cooling



At low subtransus temperatures (<850°C), the cooling rate has a minor effect on the microstructure (similar α needle sizes).

Water cooling



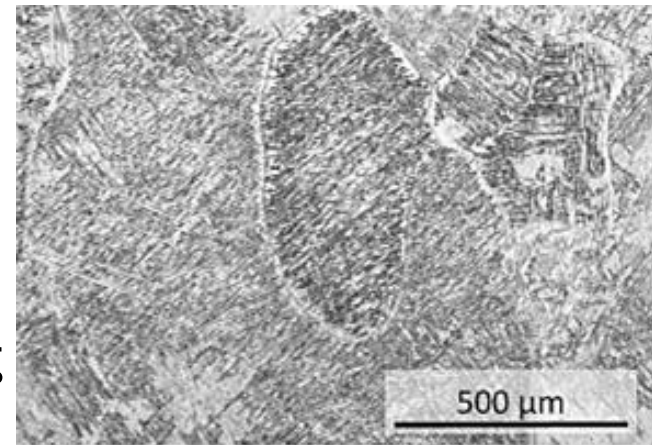
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Effect of heat treatments: influencing factors

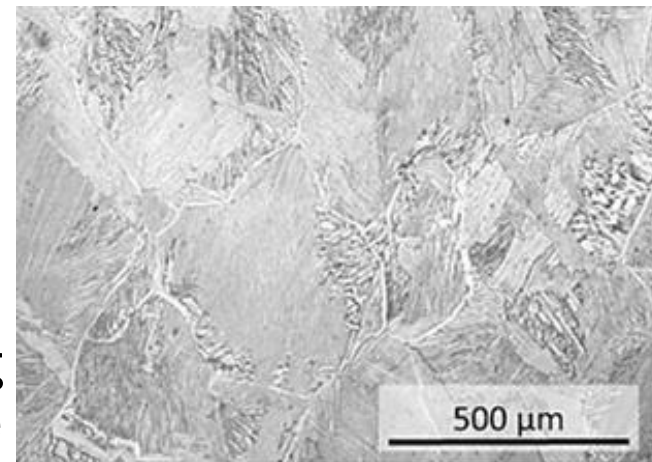
Cooling rate

At temperatures **above the transus**, the cooling rate is the most important parameter. It **determines the size and morphology of the α phase**.

Air cooling
Very fine α lamellae



Furnace cooling
Large colonies of coarse α lamellae



Thermal post-treatments of Ti-6Al-4V

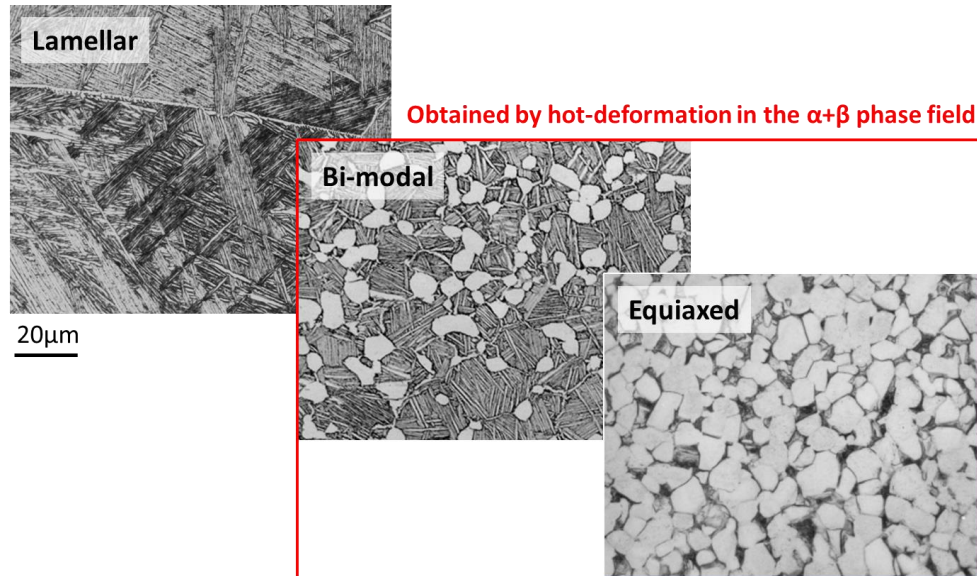
SR heat treatments have a limited influence on the microstructure.

The following features are maintained after SR:

- columnar β grain morphology: **anisotropy**
- fine lamellar microstructure: high strength, **low ductility**

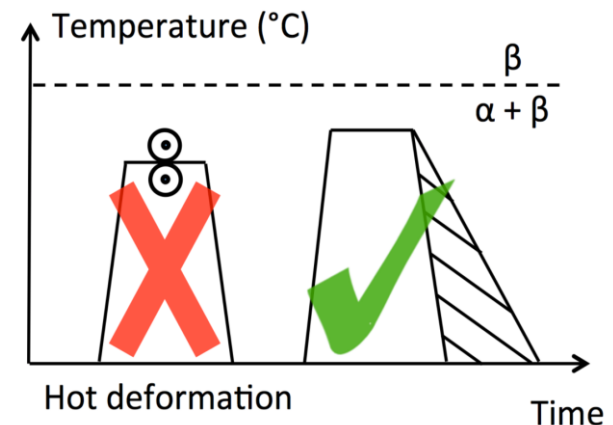
Conventional metallurgy of Ti

Hot deformation is used to obtain a microstructure with good mechanical properties (i.e. bi-modal or equiaxed).



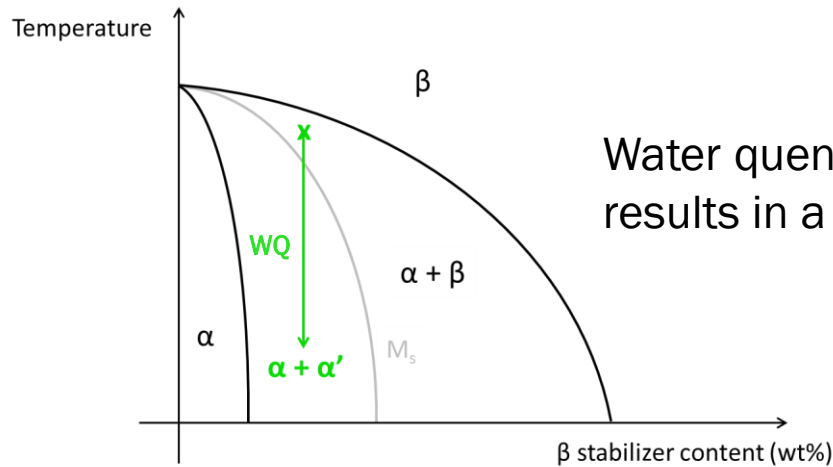
Post-processing of AM Ti

Major limitation for microstructural control: hot deformation cannot be performed on near-net shape AM parts.



Thermal post-treatments of Ti-6Al-4V

Engineering new heat treatments to tailor the microstructure of AM Ti64



Water quenching from the $\alpha + \beta$ domain (850 °C - 995 °C) results in a $\alpha + \alpha'$ microstructure.

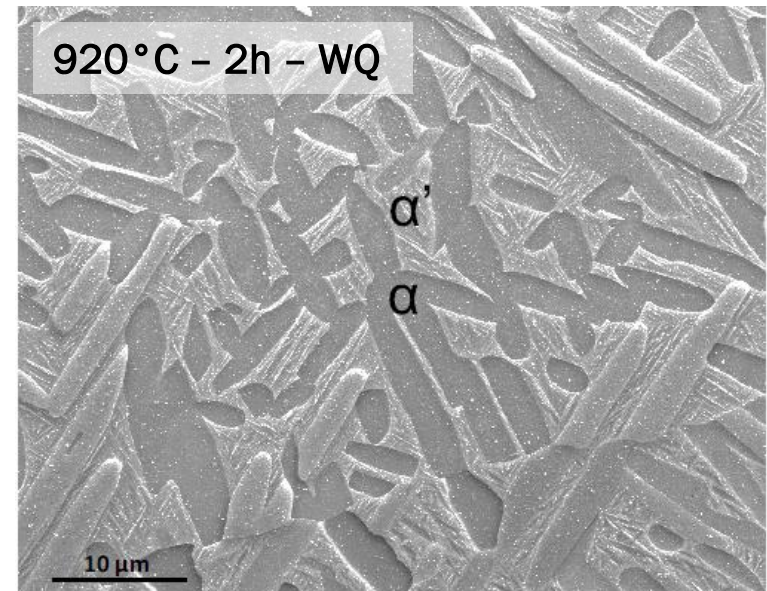
Example: 2h at 920 °C + water quenching WQ

At 920 °C: 50% β + 50% α

↓ WQ ↓

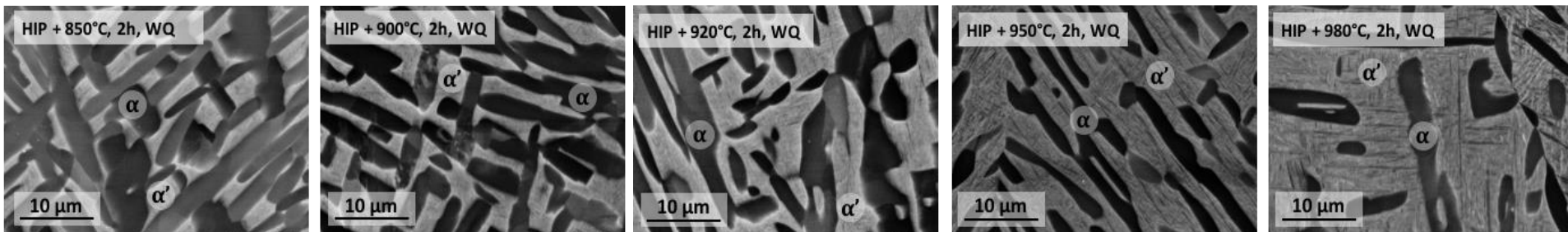
At 20 °C: 50% α' + 50% α

The “**dual-phase**” $\alpha + \alpha'$ microstructure exhibits a combination of **high strength** and **ductility**.



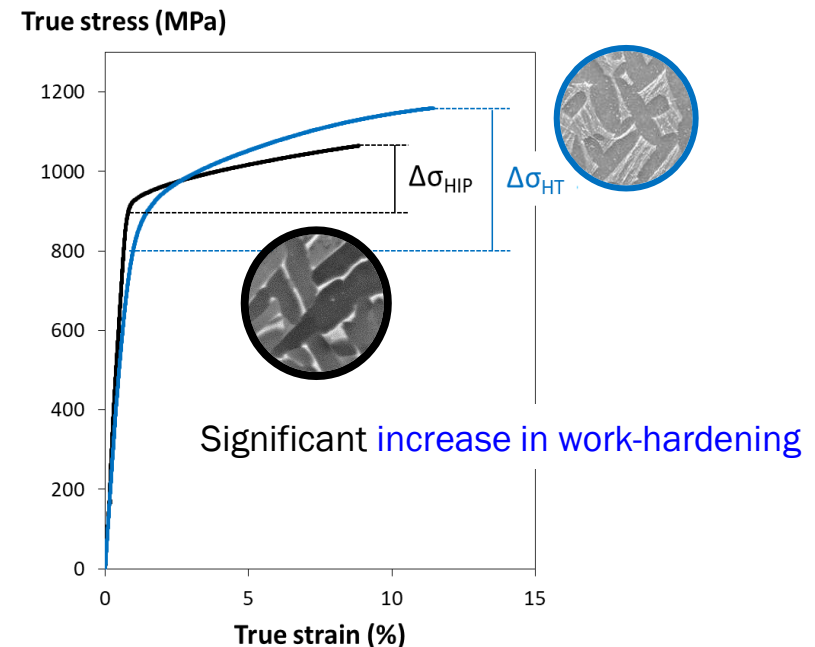
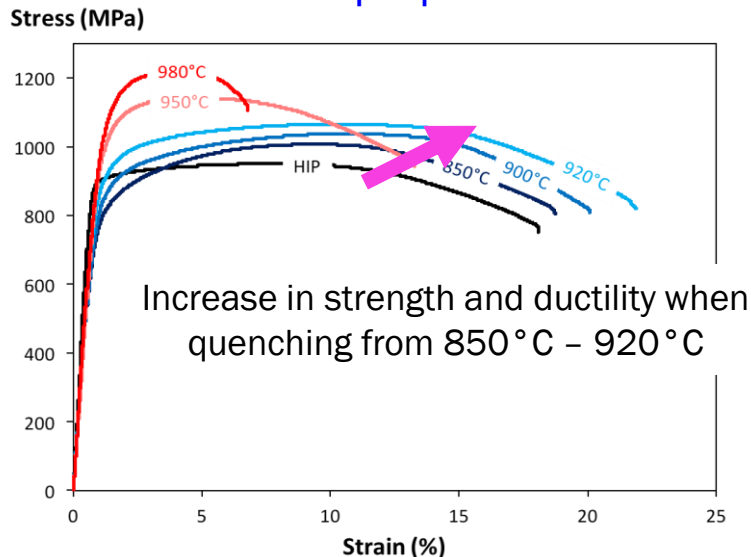
Thermal post-treatments of Ti-6Al-4V

Engineering new heat treatments to tailor the microstructure of AM Ti64



Depending on the temperature, various fractions of α' martensite are obtained.

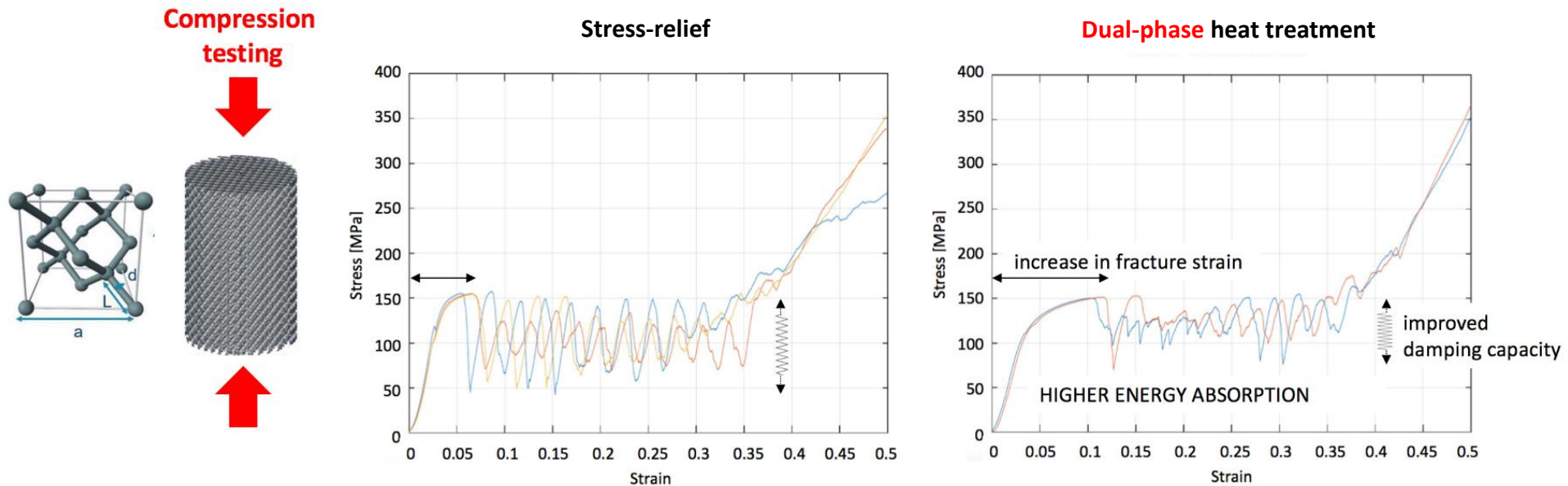
This results in a broad range of tensile properties.



Thermal post-treatments of Ti-6Al-4V

Engineering new heat treatments to tailor the microstructure of AM Ti64

Ex. of application: **energy absorption** of porous structures

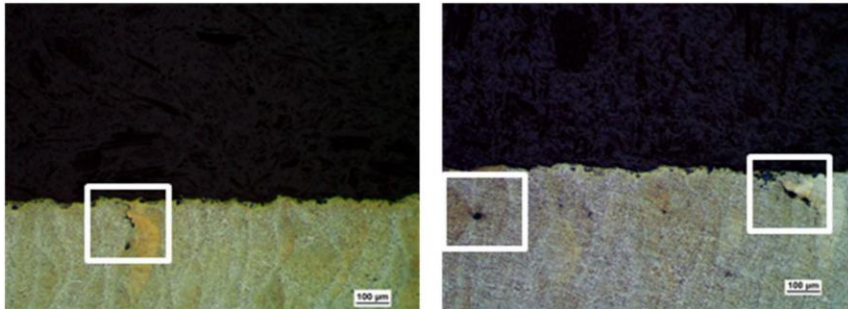
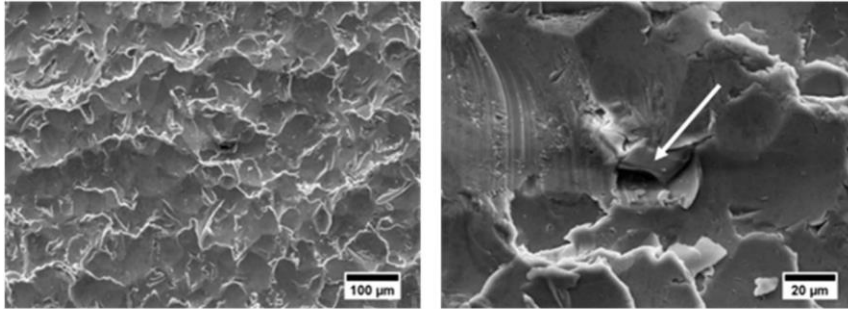


Post-treatments for microstructure and properties control

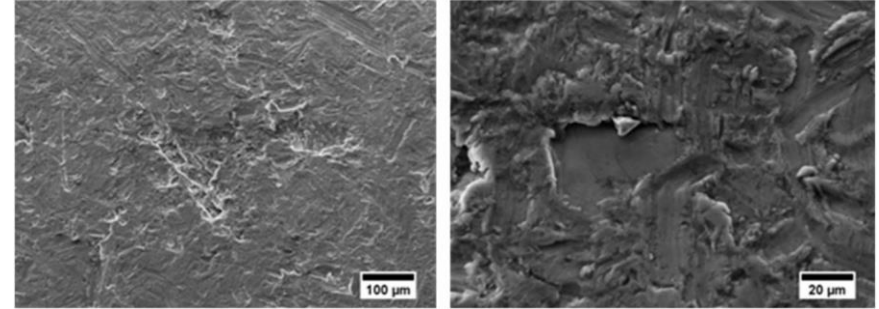
- HIP
 - process conditions
 - limitations
 - microstructural effects
- Stress-relief
 - process conditions
 - microstructural effects
- Thermal post-treatments for microstructure control
 - typical microstructural effects on other alloys
 - influencing factors (illustrated on Ti-6Al-4V)
- **Surface post-treatments**
- Effect of post-processing on mechanical properties
 - tensile
 - fatigue

Surface post-treatments

Sand blasting vs vibro-finishing



Sand blasting



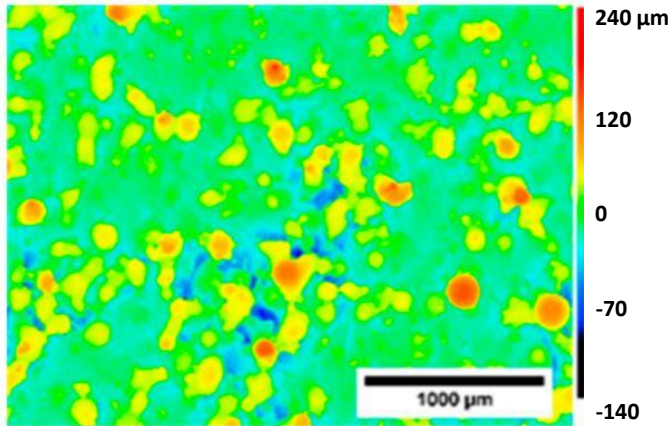
Vibro-finishing

Open pores can still be identified on the surface profile.

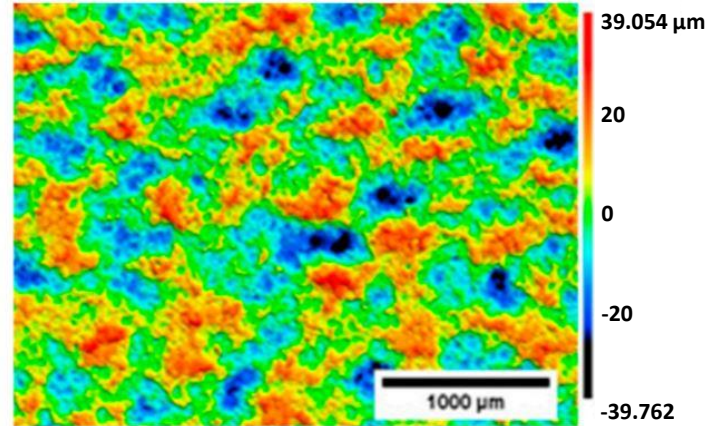
⇒ inherited from surface or subsurface defects formed during L-PBF processing

Surface post-treatments

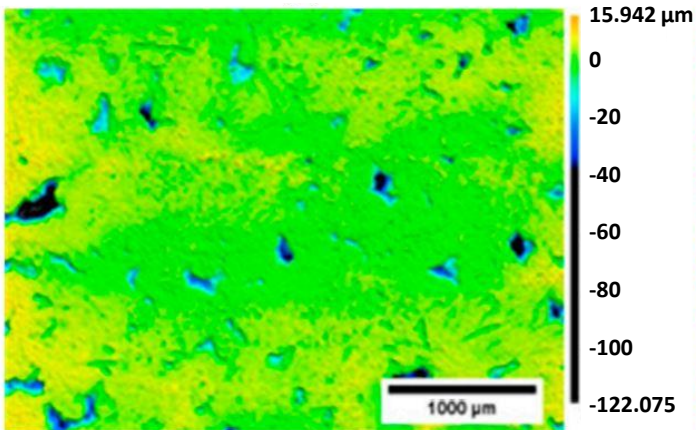
Sand blasting vs vibro-finishing



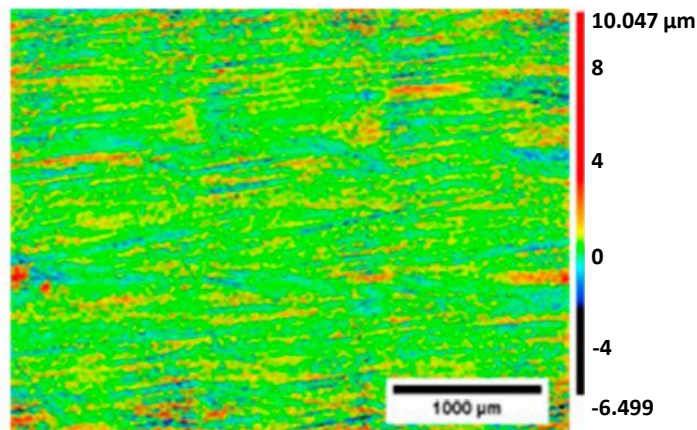
As-built



Sand-blasted



Vibro-finished



Machined and polished

Surface post-treatments

Sand blasting vs vibro-finishing

Surface Condition	S_a (μm)	S_v (μm)	S_{sk}
As-Built	15.4	85	0.74
SB	8.3	40	-0.13
VF	2.3	121	-7.31
MP	0.5	6	0.05

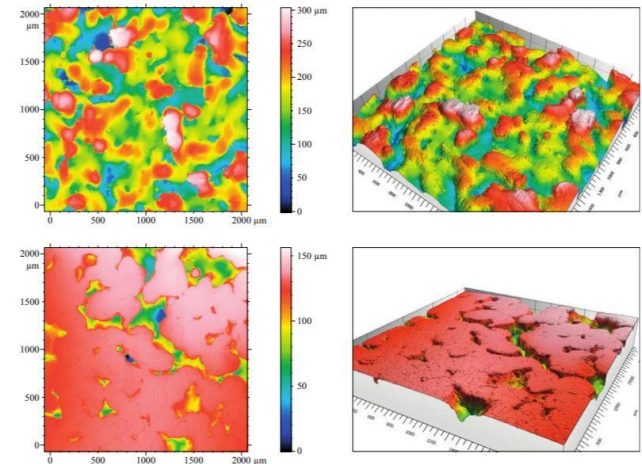
S_a arithmetical mean height

S_v maximum valley depth

S_{sk} skewness

Sand-blasting (SB) only partially reduces surface roughness, whereas **vibro-finishing** (VF) and **machining+polishing** (MP) are more effective in producing smooth surface

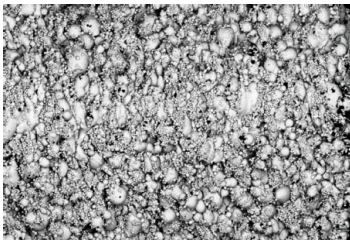
VF specimens contain **deeper and more numerous valleys**, as indicated by the negative skewness and higher S_v value



Surface post-treatments

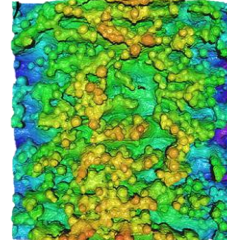
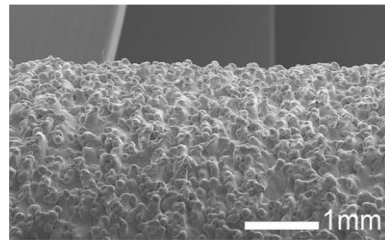
Post-process treatment: **machining vs chemical etching**

As-built (LPBF)



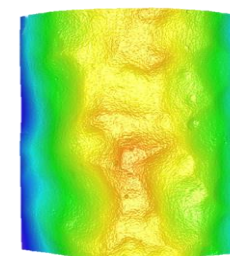
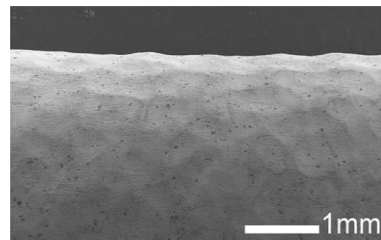
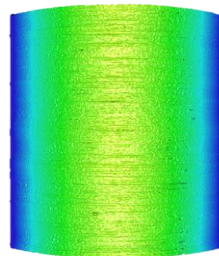
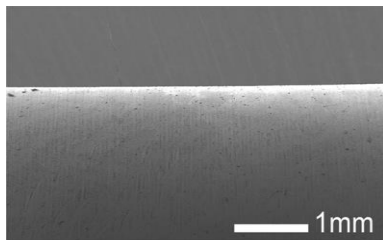
— 300 μm

As-built (EBM)



↓ machining

↓ chemical etching

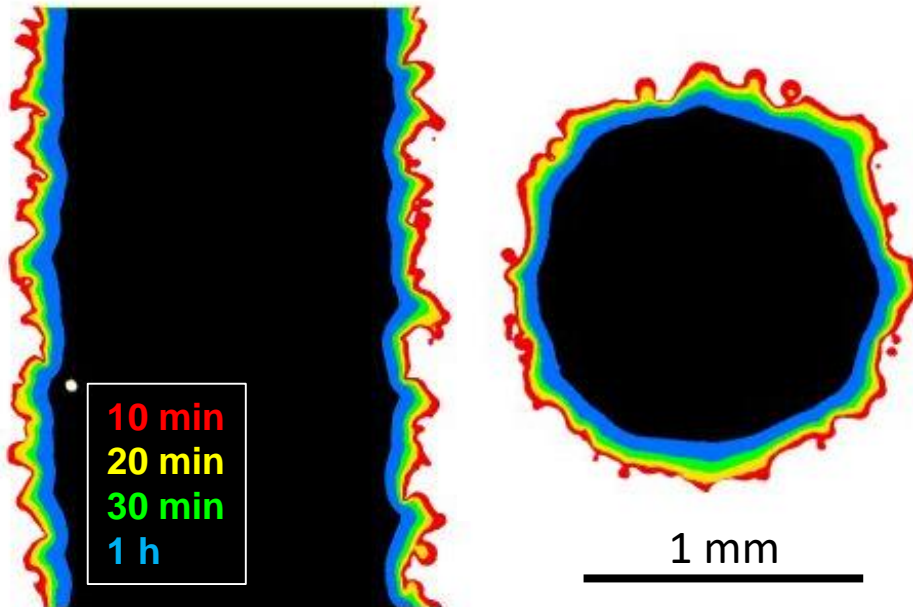


Chemical etchant:
3%HF, 13% HNO_3

Surface post-treatments

Post-process treatment: **chemical etching**

Effect of etching time:

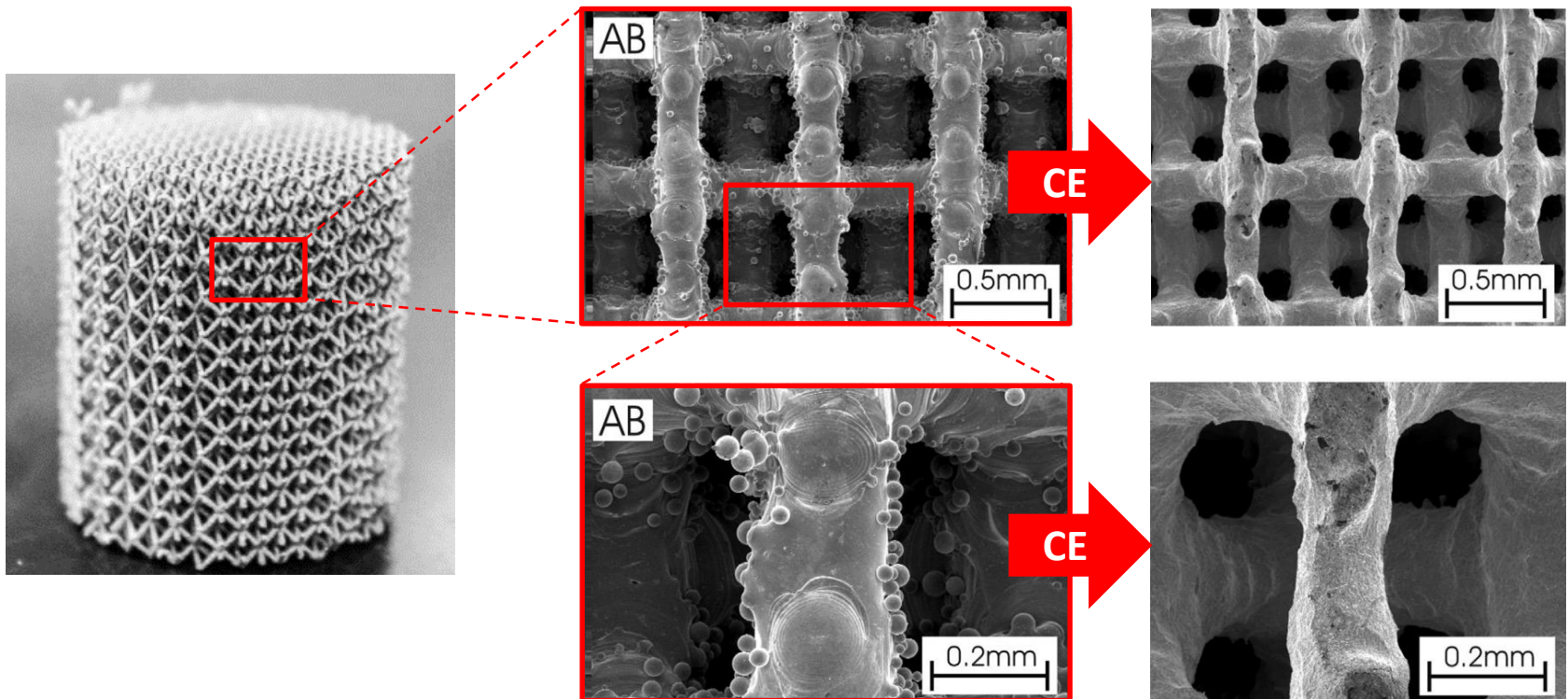


The grains of powder stuck to the surface are the first to be removed.

Surface post-treatments

Post-process treatment: **chemical etching**

Chemical etching is particularly adapted to intricate geometries (lattice structures)



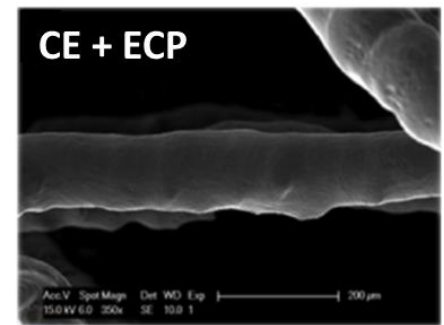
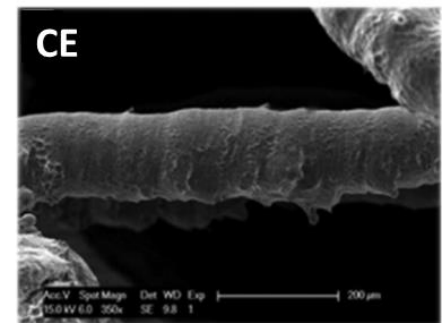
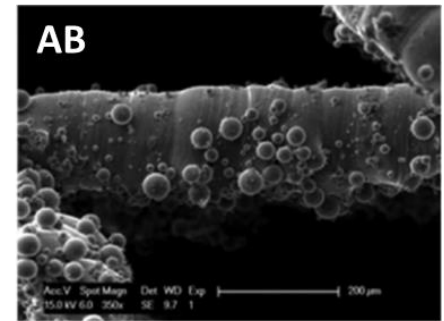
Surface post-treatments

Post-process treatment: **combination of etching and electrochemical polishing**

Chemical **etching** is mainly used **to remove** the partially melted or partially sintered **particles** on the surface.

Then, **electrochemical polishing** is used to make the surface of the part **more uniform and smooth**.

CE + ECP leads to a **significant loss in strut thickness**, which should be taken into account anticipatively in the design of the structure.

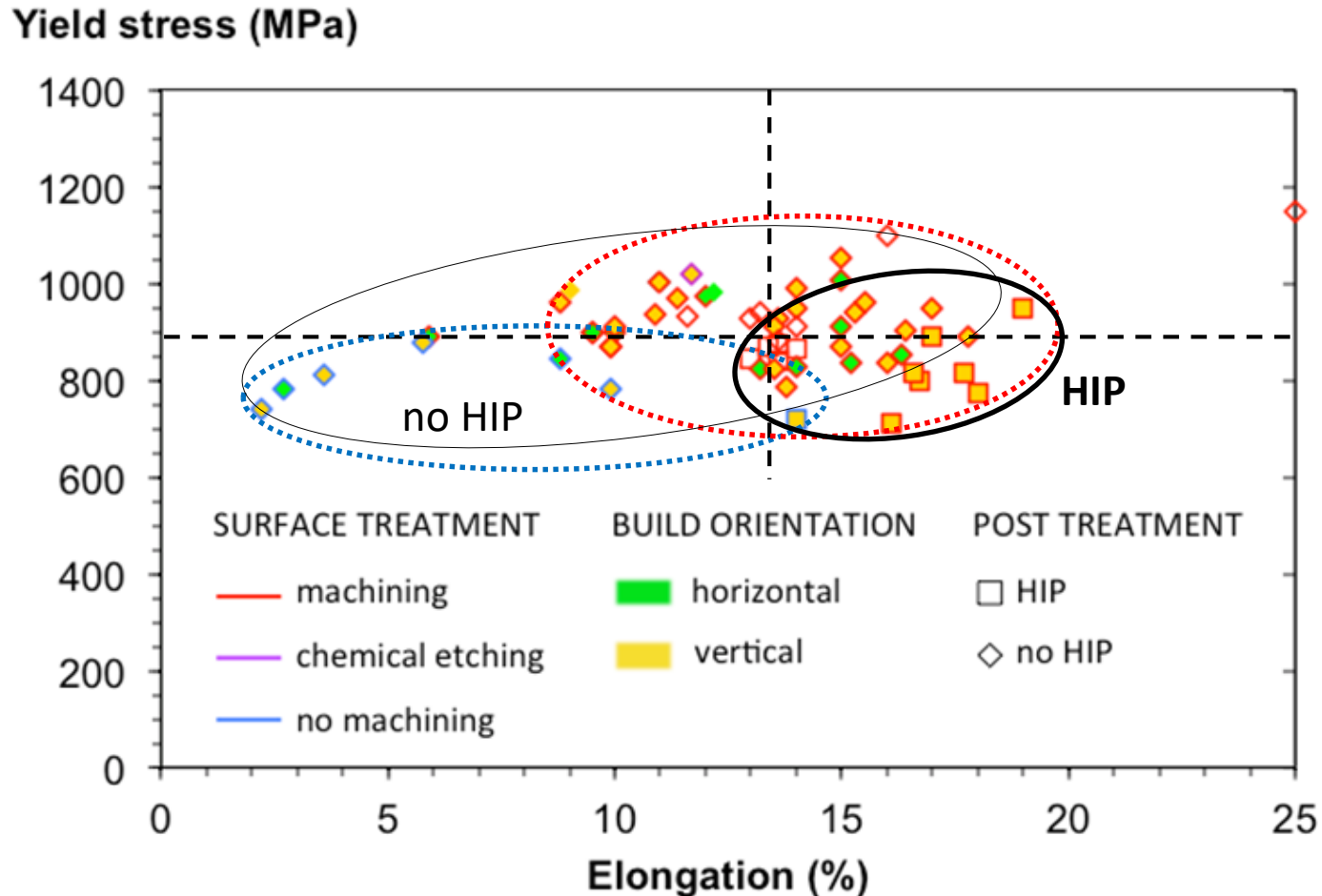


Post-treatments for microstructure and properties control

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Effect of post-processing on tensile properties

EBM Ti-6Al-4V



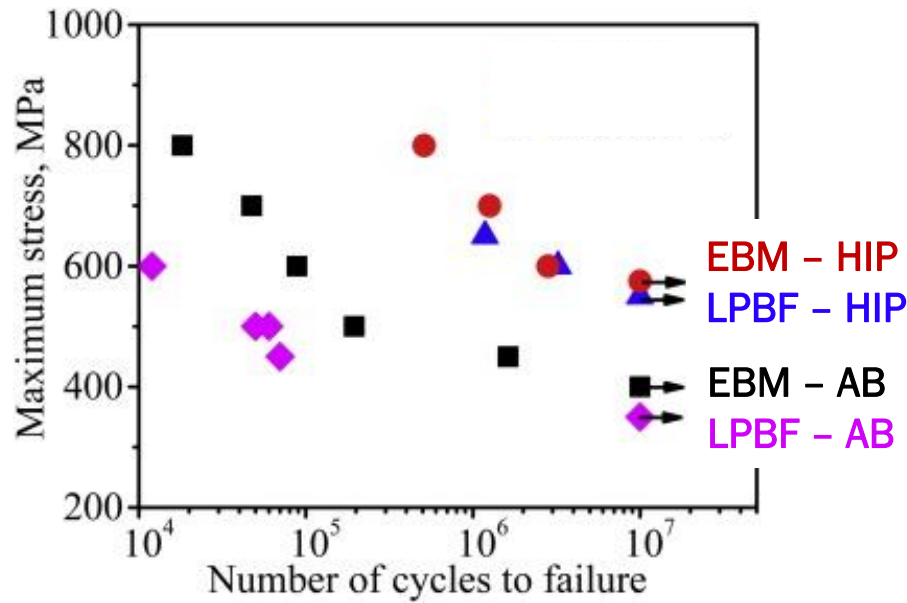
LPBF Ti-6Al-4V



Post-treatments for microstructure and properties control

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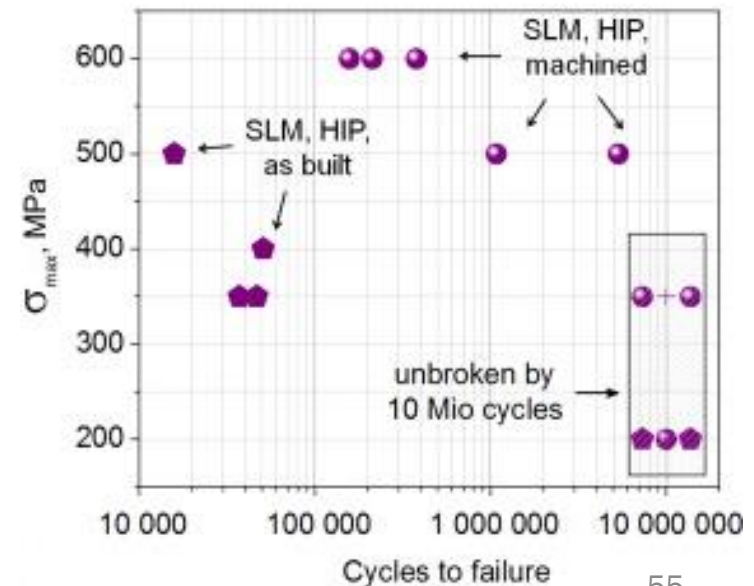
Effect of post-processing on fatigue properties of AM Ti-6Al-4V



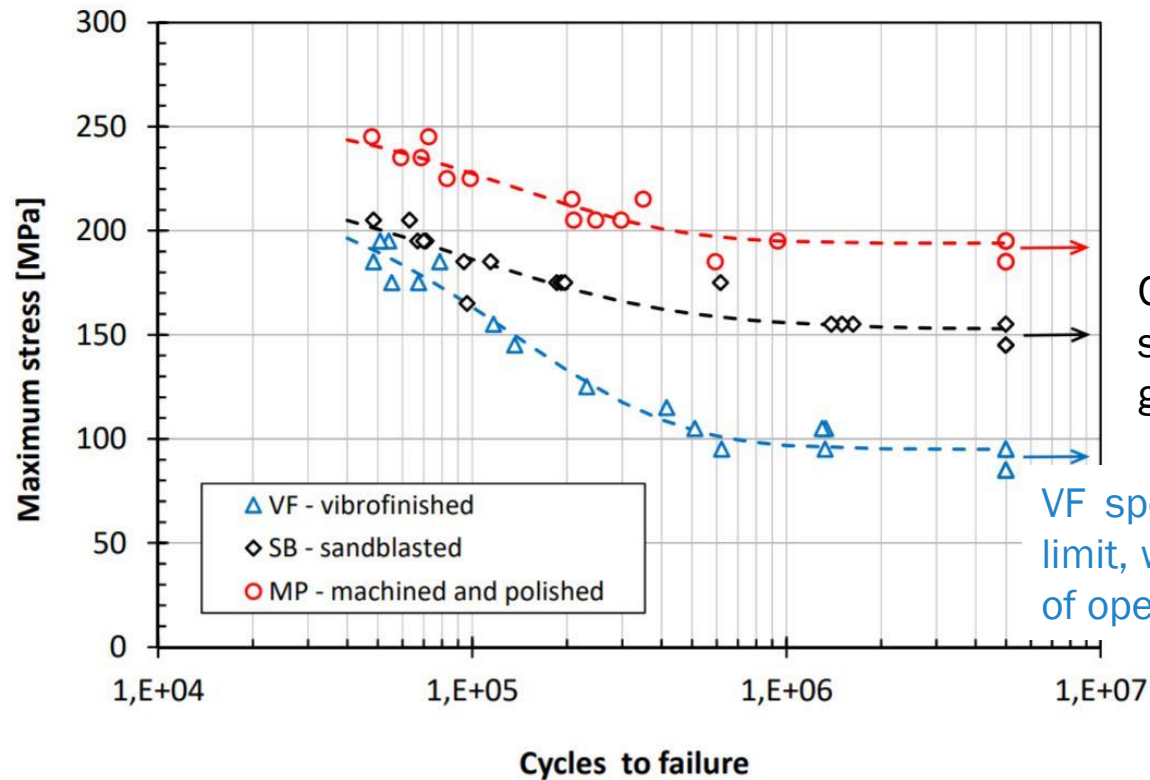
In as-built (AB) condition, LPBF parts have a lower fatigue life than EBM parts, due to high residual stresses.

HIP relieves residual stresses and removes critical defects from the bulk, increasing the fatigue life of LPBF and EBM parts.

Machining removes surface defects acting as crack initiators, further increasing the fatigue life of the material.



Effect of post-processing on fatigue properties of AM Ti-6Al-4V



Compressive residual stresses and strain hardening of the surface are generated during SB

VF specimens show the lowest fatigue limit, which is attributed to the presence of open surface and subsurface defects

